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# Polytypism and microstructures of the mixed-layer member $B_2S$ , $CaCe_3(CO_3)_4F_3$ in the bastnaesite-(Ce)–synchysite-(Ce) series

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# Abstract

The crystal structures and polytypism of bastnaesite-(Ce) (CeCO<sub>3</sub>F, B layer) and synchysite-(Ce) (CeCO<sub>3</sub>F·CaCO<sub>3</sub>, S layer) mineral series are very complicated in Sichuan Province, Southwest China. Four new polytypes: 6R,  $2H_2$ , 4H and 12H of the mixed-layer member CaCe<sub>3</sub>(CO<sub>3</sub>)<sub>4</sub>F<sub>3</sub> (B<sub>2</sub>S) of this series were discovered by X-ray diffraction analyses and high resolution transmission electron microscopy, structural symmetry, cell and subcell parameters, possible space group, and structural stacking mode (*dededgfgl...*). They reveal that the mixed-layer member B<sub>2</sub>S is formed by the ordered stacking of two structural unit layers of bastnaesite-(Ce) (B layer) and one unit layer of synchysite-(Ce) (S layer) along the *c* direction. The forming of the new polytypes may depend on distributions and the periodical variation of orientation of the [CO<sub>3</sub>] groups between the unit layers of B<sub>2</sub>S. We have observed the {0001} microtwin of the 6R polytype and the syntactic intergrowths among the different polytypes formed by stacking faults in B<sub>2</sub>S. © 2002 Elsevier Science B.V. All rights reserved.

Keywords: bastnaesite-synchysite; polytype; mixed-layer; X-ray diffraction analysis; HRTEM; transmission electron microscopy

# 1. Introduction

Bastnaesite-(Ce), Ce(CO<sub>3</sub>)F and synchysite-(Ce), CaCe(CO<sub>3</sub>)<sub>2</sub>F are the two end-members of the calcium rare-earth (Ca-RE) fluorocarbonate mineral series, constituting B layers and S layers of this polytypic mineral group. Donnay et al. [1] originally proposed that other minerals of this series can be characterized in terms of these two end-members, for example parisite-(Ce)  $[CaCe_2(CO_3)_3F_2]$  as BS and roentgenite-(Ce)  $[Ca_2Ce_3(CO_3)_5F_3]$  as BS<sub>2</sub> [1,2]. Minerals of this series are often affected by stacking faults, disorder of chemical composition, superstructures, twinning, polytypism and syntactic intergrowths during the course of crystallization and crystal growth along with other mineral phases, making it difficult to obtain crystallographic data on the fine structures of the minerals. Van Landuyt et al. discovered three new mixed-layer compounds (B<sub>3</sub>S<sub>2</sub>, B<sub>3</sub>S<sub>4</sub> and BS<sub>4</sub>) by transmission electron mi-

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croscopy (TEM) [3]. Later investigations have found 23 more  $B_mS_n$  new mixed-layer members [4–7] and 17 new polytypes of both parisite-(Ce) [8–10] and  $B_mS_n$  new mixed-layer members [5,11,12].

The microstructures of members of the Ca-RE fluorocarbonate mineral series occurring within an alkali aegirine granite massif in Sichuan Province, Southwest China, have been studied in detail by single-crystal X-ray diffraction (XRD) analysis. Areas were selected to obtain electron diffraction (SAED) and high resolution transmission electron microscope (HRTEM) data, revealing four new polytypes of mixed-layer member  $CaCe_3(CO_3)_4F_3$  (B<sub>2</sub>S) and also confirming three polytypes  $(2H_1, 12R \text{ and } 24R)$  of B<sub>2</sub>S found by Wu et al. using the crystal lattice imaging technique [4,11,12]. The investigation of the crystallographic system, cell parameters, chemical formula, space group, structural stacking mode and microtwin as well as syntactic intergrowth between polytypes of  $B_2S$  is described.

## 2. Samples and experimental technique

The samples for this investigation were crystal grains of Ca-RE fluorocarbonate minerals (0.08-0.3 mm in diameter) from a rare-earth mineral deposit within the Sichuan alkali aegirine granite massif. 'Polycrystals' with syntactic intergrowths were selected for TEM analysis after single-crystal XRD analyses, crushed into fine fragments and suspended in absolute alcohol. A drop of the suspension was put on a copper grid coated with a perforated carbon film, in turn coated with gold. It was examined at 200 kV using a JEOL-2000EX II HRTEM equipped with a top-entry goniometer stage ( $\pm 10^{\circ}$  tilt) and an ultra-high-resolution pole piece (Cs = 0.7 mm) with an interpretable point resolution of 0.21 nm. SAED patterns at 120 kV on a Philips-CM 12 ( $\pm 45^{\circ}$  tilt, Cs = 2.0 mm) TEM.

The single crystals of a mixed-layer member  $(B_2S)$  discovered by XRD analysis are columnar, 0.1–0.3 mm in diameter with transverse striations on the crystal surface (Fig. 1). They are yellow or off-white in color, transparent, with glassy or

greasy luster and few cleavages, similar optical properties to those of parisite-(Ce) [13]. X-ray single-crystal diffraction analyses were performed by using an RU-200B single-crystal diffractometer with rotation angles of 30–60° with CuK $\alpha$  radiation ( $\lambda$ =0.15418 nm) and no filter at 40 kV, 100 mA and 2–4 h exposure.

## 2.1. X-ray single-crystal diffraction studies

Rotation diffraction patterns of the mixed-layer member  $(B_2S)$  about the *c* axis of single crystals (Fig. 2) were observed in X-ray single-crystal diffraction studies. Local enlarged patterns (Fig. 3) were obtained from the diffraction spot array between 11  $\overline{2}0$  and 11  $\overline{2}L_1$ " in the rotation diffraction pattern around the c axis of Ca-RE fluorocarbonate minerals (B, B<sub>2</sub>S, BS, BS<sub>2</sub> and S). Donnay et al. [1] have previously described the B, BS, BS<sub>2</sub> and S XRD patterns. In Figs. 2 and 3 crucial differences in diffraction intensity, repeat period and distributional positions of diffraction spots in the spot array (except for the strongest array 11  $\overline{20}$ , 11  $\overline{2}L_1'''$ , 11  $\overline{2}L_2'''$ , ...) can be seen between the mixed-layer member (B<sub>2</sub>S) and the other known Ca-RE fluorocarbonate minerals [1,13]. They can be summarized as follows:

- 1. In the rotation diffraction pattern (Fig. 2) about the *c* axis, the diffraction spot array H-K=3n (n=0, 1, 2,...) and the strongest spots 11  $\overline{2}L_i'''$  ( $i=0, \pm 1, \pm 2,...$ ) reveal the existence of pseudo-periods *a'* and *c'''* (*c'''*=0.473 nm).
- 2. The distribution between strong diffraction spots and weak ones is similar to that of parisite-(Ce) and synchysite-(Ce), but the distributional positions and repeat period of the diffraction spots are different. There are three stronger diffraction spots between the strongest ones, 11  $\overline{20}$  and 11  $\overline{2}L_1'''$ , revealing a subperiod c'' (c'' = 4c''' = 1.892 nm).
- In the spot array H−K=3n between the strongest and stronger spots, every four weak spots representing the subperiod c' (c' = 2c'' = 8c''' = 3.785 nm) are distributed alternately with three stronger ones. Very weak diffraction spots H−K≠3n are scattered at 1/3 or 2/3 the distance between two adjacent arrays close to the

zero array, so the superstructural period is c = 3c' = 6c'' = 24c''' = 11.354 nm. Thus the new polytype (6*R*) of B<sub>2</sub>S is rhombohedral (Table 1).

4. By using the Donnay et al. [1] plot of the pseudo-period c<sup>m</sup> obtained by X-ray single-crystal diffraction analysis against composition in mol% CaCO<sub>3</sub>, the proportion of CaCO<sub>3</sub> can be inferred. c<sup>m</sup> = 0.473 nm, giving 25 mol% CaCO<sub>3</sub>, i.e. Ce(CO<sub>3</sub>)F:CaCO<sub>3</sub> = 3:1. The chemical formula of this mixed-layer member (B<sub>2</sub>S) is therefore CaCe<sub>3</sub>(CO<sub>3</sub>)<sub>4</sub>F<sub>3</sub>, composed of two formula units of bastnaesite-(Ce) (2Ce(CO<sub>3</sub>)<sub>2</sub>F) and one of synchysite-(Ce) (CaCe(CO<sub>3</sub>)<sub>2</sub>F). It represents a mixed-layer member BBS = B<sub>2</sub>S intermediate between bastnaesite-(Ce) and parisite-(Ce).

# 2.2. HRTEM observations and discussion

The HRTEM specimens were carefully selected after X-ray single-crystal diffraction analysis. The crystal appearance and other characteristics are similar to those of parisite-(Ce), but in fact most of the specimens were 'polycrystals', composed of interlayers of bastnaesite-(Ce) [B layer, CeF-CO<sub>3</sub>], parisite-(Ce) (BS layer), roentgenite-(Ce) (BS<sub>2</sub> layer) and synchysite-(Ce) [S layer, CeF-CO<sub>3</sub>-Ca-CO<sub>3</sub>]. The crystal structures characteristic of the

Table 1 Crystal structures of Ca-RE fluorocarbonate mineral series

Ca-RE fluorocarbonate minerals are rhombic planes constituted of heavy atoms of Ce and Ca distributed alternately normal to the trigonal or hexagonal axis, and triangular planes formed by  $CO_3$  lying parallel to the *c* axis between the CeF and CeF layers, or oblique from the c axis between the CeF and Ca layers. Donnay et al. [1] made qualitative descriptions of this mineral using four different layers parallel to the (0001) plane. The four layers are: d, CeF ionic layers; f, Ca ionic layers: e, layers of CO3 groups between CeF layers, and g, layers of CO<sub>3</sub> groups between Ca layer and CeF layer. For example, the stacking layers of bastnaesite-(Ce) (B), parisite-(Ce) (BS), roentgenite-(Ce)  $(BS_2)$  and synchysite-(Ce) (S) can be described as de (B), dedgfg (BS), dedgfgdgfg (BS<sub>2</sub>) and dgfg (S).

# 2.3. The 6R polytype of $B_2S$ and its microtwin

A set of SAED patterns (Fig. 4a–c) were found around the 000*l* spot array where the crystal zones are  $[11\ \overline{2}0]$  (Fig. 4a),  $[01\ \overline{1}0]$  (Fig. 4b) and  $[\overline{1}2\ \overline{1}0]$ (Fig. 4c) respectively, and the angles between two adjacent electron diffraction patterns are 29.9° (between Fig. 4a and b) and 30.2° (between Fig. 4b and c). A reciprocal lattice plane (Fig. 4d) normal to the crystal zone [0001] was constituted according to the three SAED patterns and the

| Name                          | Space group                     | Cell      |           |           |          | Structural<br>layer<br>stacking<br>mode | Height of<br>layer | Chemical<br>formula  | CaCO <sub>3</sub> | References      |
|-------------------------------|---------------------------------|-----------|-----------|-----------|----------|---|--------------------|--|-------------------|-----------------|
|                               |                                 | a<br>(nm) | b<br>(nm) | c<br>(nm) | β<br>(°) |   | (nm)               |  | (mol%)            |                 |
| Bastnaesite-(Ce)              | <i>P</i> 6 2c                   | 0.716     | 0.716     | 0.978     | 120.00   | del                                     | 0.489              | CeCO <sub>3</sub> F  | 0.00              | [1,3,4,14]      |
| $\mathbf{B}_2\mathbf{S}(12R)$ | R32 or R3m or $R \overline{3}m$ | 0.711     | 0.711     | 22.608    | 120.00   | dededgfg/                               | 1.892              | $CaCe_3$<br>(CO <sub>3</sub> ) <sub>4</sub> F <sub>3</sub>   | 25.00             | [4,11,12]       |
| Parisite-(Ce)                 | C2/c or Cc                      | 1.231     | 0.711     | 2.825     | 98.26    | dedgfg/                                 | 1.413              | $CaCe_2$<br>(CO <sub>3</sub> ) <sub>3</sub> F <sub>2</sub>   | 33.00             | [8]             |
| Parisite-(Ce)                 | R3 or $R\overline{3}$           | 0.717     | 0.717     | 8.478     | 120.00   | dedgfg/                                 | 1.413              | $CaCe_2$<br>(CO <sub>3</sub> ) <sub>3</sub> F <sub>2</sub>   | 33.00             | [1,3,4,9–12,16] |
| Roentgenite-(Ce)              | R3 or $R\overline{3}$           | 0.713     | 0.713     | 6.940     | 120.00   | dedgfgdgfg/                             | 2.313              | $Ca_2Ce_3$<br>(CO <sub>3</sub> ) <sub>5</sub> F <sub>3</sub> | 40.00             | [1–3]           |
| Synchysite-(Ce)               | C2/c                            | 1.233     | 0.711     | 1.874     | 102.68   | dgfg/                                   | 0.937              | CaCe<br>(CO <sub>3</sub> ) <sub>2</sub> F                    | 50.00             | [1,15]          |



Fig. 1. Scanning electron micrograph of a single crystal of  $B_2S$ .

angles between two adjacent zones. In Fig. 4a–c, the diffraction spots produced by this mixed-layer member (B<sub>2</sub>S) can be classified into strong, stronger and weak ones. The diffraction spots of the zones [ $\overline{12}$   $\overline{10}$ ] and [11  $\overline{20}$ ] are characteristic of a distribution in layers parallel to the *c* direction. The 3*n* (*n*=0, ±1, ±2,...) layers are constituted by strong 000<u>24</u> (*c*<sup>*m*\*</sup>) and stronger diffraction

spots have the same arrangement, reflecting the hexagonal subcell a'c'' and a'c'' (c''=4c'''). The  $3n \pm 1$  layers constituted by weak spots whose distributional density is twice that in the 3n layers, reveal the substructure a'c' (c' = 2c'') and the superstructure ac ( $a = \sqrt{3}a'$ , c = 3c'). There are five extinction spots between 0000 and 0006, the above diffraction spots display rhombohedral (R)lattice symmetry and the reflection condition is h-k+l=3n (Fig. 4a-e and Table 2), so the first strong diffraction spot should be indexed as 00024  $(c'''^*)$  and the diffraction spot nearest to the transmission spot 0000 should be indexed as 0006 ( $c''^*$ ) along the c direction. The cell parameters obtained by SAED analysis are a' = 0.412 nm, a = 0.713 nm, c''' = 0.471 nm, c = 3c' = 6c'' =24c''' = 11.304 nm. The possible space group is R3 or  $R\overline{3}$ . The HRTEM image (Fig. 4e) of the crystal zone  $[12 \ \overline{10}]$  reveals that the period distance of the structural unit layers CeF-CO3-CeF-CO<sub>3</sub>-CeF-CO<sub>3</sub>-Ca-CO<sub>3</sub> is 1.884 nm. The unit layer is formed by the basic layer CeF-CO<sub>3</sub>-CeF-CO<sub>3</sub>-CeF (deded), which is composed of three black streaks (CeF layer), two white thin streaks CO<sub>3</sub> between the black ones and a white thick streak  $CO_3$ -Ca-CO<sub>3</sub> (gfg). The corresponding stacking mode is *dededgfg/...*. From the chemical composition  $[CaCe_3(CO_3)_4F_3]$  and the stacking mode, the mixed-layer member can be considered as BBS, i.e.  $B_2S$  (6R). The projective



Fig. 2. Single crystal XRD photograph (rotation about the *c* axis) of the mixed-layer member  $B_2S$  (CuK $\alpha$  radiation; 2 h exposure).

| Polytype                 | Cell and subcell (nm) |       |                                      |                                    | Number of<br>unit layers<br>( <i>dededgfg</i> ) | Reflection co | Possible<br>space group |         |                       |  |
|--------------------------|-----------------------|-------|--------------------------------------|------------------------------------|---|---------------|-------------------------|---------|-----------------------|--|
|                          | <i>a</i> ′            | а     | <i>c</i> ′                           | С                                  | -   | hkil          | $h \overline{h}0l$      | hh2 hl  | 0001                  |  |
| 6 <i>R</i> <sup>a</sup>  | 0.412                 | 0.713 | 3.768<br><i>c'</i> = 8 <i>c'''</i>   | 11.304<br>c = 24c'''               | 6   | h-k+l=3n      | 2h+l=3n                 | l=3n    | l = 6n                | $R3, R\overline{3}$  |
| 12 <i>R</i> <sup>b</sup> | 0.410                 | 0.711 | 7.536<br><i>c</i> ′ = 16 <i>c</i> ‴  | 22.608<br>c = 48c'''               | 12  | -h+k+l=3n     | h+l=3n                  | l = 6n  | <i>l</i> =12 <i>n</i> | $\frac{R32}{3m}, R3m, R$   |
| 24 <i>R</i> <sup>b</sup> | 0.407                 | 0.705 | 15.072<br><i>c</i> ′ = 32 <i>c</i> ‴ | 45.216<br><i>c</i> = 96 <i>c</i> ‴ | 24  | h-k+l=3n      | 2h+l=3n                 | l = 12n | l = 24n               | $R3, R\overline{3}$  |
| $2H_1^{b}$               | 0.410                 | 0.710 | -                                    | 3.768<br>c = 8c'''                 | 2   | -             | _                       | -       | l=2n                  | <i>P</i> 6 <sub>3</sub> , <i>P</i> 6 <sub>3</sub> / <i>m</i>           |
| $2H_2^a$                 | 0.411                 | 0.712 | -                                    | 3.768<br>c = 8c'''                 | 2   | -             | l=2n                    | -       | l=2n                  | P6 <sub>3</sub> /mmc,<br>P6 <sub>3</sub> mc, P <u>6</u> 2c             |
| $4H^{\rm a}$             | 0.407                 | 0.705 | -                                    | 7.536<br><i>c</i> = 16 <i>c</i> ‴  | 4   | -             | _                       | l = 2n  | l = 4n                | $P6_3/mmc$ ,<br>$P6_3mc$ , $P \overline{6}2c$                          |
| 12 <i>H</i> <sup>a</sup> | 0.405                 | 0.702 | -                                    | 22.608<br>c = 48c'''               | 12  | _             | _                       | l = 6n  | <i>l</i> =12 <i>n</i> | $\begin{array}{c} P6_3/mmc, \\ P6_3mc, P \ \overline{6}2c \end{array}$ |

Table 2 TEM analyses of seven polytypes of the mixed-layer member  $B_2 S \label{eq:eq:seven}$ 

Wu et al. [4,11,12].  $a = \sqrt{3}a'$ ; c''' = 0.471 nm; c'' = 1.884 nm; c'' (height of unit layer) = 4c'''; R: rhombohedral; H: hexagonal. <sup>a</sup> New polytype.

<sup>b</sup> Known polytype.

positions of the unit layers (c'' = 1.884 nm) of B<sub>2</sub>S along [ $\overline{12}$   $\overline{10}$ ] (Fig. 4e) were obtained using the thickness and the projective positions of the different ionic groups d, e, f and g [14,15] and the computation of the diffraction spots c'''\* (00024) and c''\* (0006) on the  $c^*$  spot array (Figs. 4a-e, 8 and 9 and Table 2).

Fig. 4f–g shows a {0001} microtwin of B<sub>2</sub>S-6*R* to be a rotation twin with rotation angle 180°, whose twin planes and twin axes are {0001} and [0001] respectively. The diffraction spots spread in

layers along the *c* direction. In the  $h\bar{h}0l$  layers, two sets of diffraction spots of matrices (*M*) and twins (*T*) coincide with each other when  $h = |\bar{h}| = 3n$  ( $n = 0, \pm 1, \pm 2,...$ ), but split into pairs, the split distance being 1/6 of the spot distance in the 3*n* layers (layers of the strong and stronger diffraction spots) or *c*"\*/6 when  $h = |\bar{h}| \neq 3n$ ( $n = 0, \pm 1, \pm 2,...$ ). The crystal zones of the matrices (*M*) and twins (*T*) in the {0001} microtwin of B<sub>2</sub>S (6*R*) were indexed as [11  $\overline{2}0$ ]<sub>*M*</sub> and [ $\overline{1}$  $\overline{120}$ ]<sub>*T*</sub>.



Fig. 3. Schematic representations of the  $11 \overline{2}0-11 \overline{2}L_1^m$  diffraction spots in XRD rotation patterns about the *c* axis of bastnaesite (B), B<sub>2</sub>S, parisite (BS), roentgenite (BS<sub>2</sub>) and synchysite (S).





Fig. 5. SAED pattern and HRTEM image of the  $B_2S-2H_1$  polytype corresponding to zones [12 10].

# 2.4. Three 2nH polytypes of $B_2S$

The polytypism of this mineral series is complex [1,8–12,16]. Four 2nH polytypes (2H<sub>1</sub> [11,12], 2H<sub>2</sub>, 4H and 12H) of B<sub>2</sub>S and three 3nR polytypes (6R, 12R [4,11,12] and 24R [11]) have been found in the 'polycrystals'. The diffraction spots in the 3n layers have the same arrangement in the SAED patterns of zones [12 10], [11 20] and [2 1 10], revealing the same subcell dimensions a'c''' and a'c'' (c'' = 4c'''). The distribution of the weak diffraction spots in the  $3n \pm 1$  layers is different and their density reveals different polytypic sequences of B<sub>2</sub>S.

The four 2nH ( $2H_1$  [11,12],  $2H_2$ , 4H and 12H) polytypes of B<sub>2</sub>S are illustrated in Figs. 5, 6 and 7a,b. Their subcells (a'c'' and a'c'') and supercells (a c) all have hexagonal symmetry with a *P* lattice (Table 2 and Fig. 9). This is revealed by the distributional characteristics of the diffraction spots in the SAED patterns of zones [12 10] or [11 20] and the structural stacking mode in the HRTEM images. For example, the polytype B<sub>2</sub>S-2H<sub>1</sub> [11,12] was recognized from the SAED and HRTEM analyses shown in Fig. 5.

Fig. 6a-d shows the SAED patterns of the crystal zones  $[01 \ \overline{10}], [\overline{12} \ \overline{10}], [\overline{45} \ \overline{10}]$  and  $[\overline{1100}]$  in which the angles between each two adjacent zones are 29.7° (Fig. 6a,b), 19.3° (Fig. 6b,c) and 10.6° (Fig. 6c,d), respectively. In the SAED pattern of the zone  $[12 \ \overline{10}]$ , the density of the diffraction spots in the  $3n \pm 1$  layers and the 3n layers is the same, but for  $B_2S-2H_1$ , the density of the diffraction spots in the  $3n \pm 1$  layers is twice that in the 3n layers in Fig. 5. The spot distribution is H-lattice symmetrical (see Fig. 6a-d), so the diffraction spot nearest to the spot 0000 in the 000l spot array has the index 0002. In the case of zone  $[\overline{12} \ \overline{10}]$  (Fig. 6e), the corresponding HRTEM image reveals an arrangement of different ionic layers and repeated periodical stacking of structural unit layers (dededgfg, c"=1.884 nm) along the c direction of the polytype  $B_2S-2H_2$ . Thus the cell height is made up of two unit layers (dededgfg), i.e.  $c = 2c'' = 2 \times 1.884$  nm = 3.768 nm, and it can be called a two-layer polytype. Fig. 6f also shows an SAED pattern and HRTEM image of zone [1100] of polytype  $B_2S-2H_2$ . Two polytypes 4H (a four-layer polytype) and 12H (a twelve-layer polytype) of  $B_2S$  are shown in Figs.

<sup>←</sup> 

Fig. 4. (a–c) SAED patterns of B<sub>2</sub>S (6*R*) obtained by tilting the crystal about [0001]. (d) A diagram of the reciprocal lattice plane determined from the SAED patterns a–c. Triangles and circles represent the spots projected onto the reciprocal lattice plane (0001)\* from 1/6  $d_{(0006)}$  and 2/6  $d_{(0006)}$  above, respectively. (e) HRTEM image of B<sub>2</sub>S (6*R*) corresponding to zone [12 10]. (f,g) SAED pattern and indexing of {0001} microtwinning in B<sub>2</sub>S (6*R*), corresponding to zone [11 20]. Spots and circles represent the matrix (*M*) and twin (*T*).



Fig. 6. (a-d) SAED patterns of the  $2H_2$  polytype obtained by tilting the crystal about [0001]. (e,f) SAED patterns and HRTEM images of the B<sub>2</sub>S-2H<sub>2</sub> polytype corresponding to zones [12 10] (e) and [1100] (f).



Fig. 7. SAED patterns of the  $B_2S-4H$  (a) and  $B_2S-12H$  (b) polytypes corresponding to zone [12 10].

7a,b and 9 and Table 2. They have been identified by similar methods to  $B_2S-2H_2$ .

In the cases of the three 3nR polytypes of B<sub>2</sub>S (6*R*, 12*R* [4,11,12] and 24*R* [11]), the smallest subcells (*a'c'''*) and subcells (*a'c''* and *a'c'*) have hexagonal symmetry, while the supercells (*ac*) are rhombohedral. In the SAED patterns of the zones [12 10], [11 20] and [2 1 10], the reflection law of the diffraction spots is *hkil*, -h+k+l=3n (12*R*) or h-k+l=3n (6*R* and 24*R*) (Table 2, Figs. 4 and 9). In Fig. 4, there are five extinction spots between the transmission spot 0000 and stronger diffraction spot 0006, corresponding to a 6R (six-layer) polytype of B<sub>2</sub>S.

# 2.5. Syntactic intergrowths between polytypes of $B_2S$

Syntactic intergrowths are a significant feature of the Ca-RE fluorocarbonate mineral series. Syntactic intergrowths among the three polytypes  $2H_1$ , 6R and 4H of B<sub>2</sub>S have been observed in



Fig. 8. Schematic representations of the crystallo-structural stacking modes of bastnaesite, the mixed-layer member  $B_2S$ , parisite, roentgenite and synchysite.



Fig. 9. Schematic representations of  $[12 \ \overline{10}]$  SAED patterns of five polytypes in  $B_2S$ .

this study. An SAED pattern of the zone [11  $\overline{20}$ ] (Fig. 10b) has only one set of B<sub>2</sub>S diffraction spots (c'' = 1.884 nm) along the *c* direction, whereas there are three sets produced by the 2*H*<sub>1</sub>, 6*R* and 4*H* polytypes superposed mutually along the  $h\overline{h}0l$  spot array ( $h = |\overline{h}| = 3n \pm 1$ , n = 0,  $\pm 1$ ,  $\pm 2$ ,...). A corresponding enlarged local pattern and the results of indexing are shown in

Fig. 10a, where the stronger diffraction spots nearest to the spot 0000 are produced by the lattice planes (0002)  $(2H_1)$ , (0006) (6R) and (0004) (4H). Further research reveals that the syntactic intergrowths among different polytypes are due to a disordered stacking structure formed by the stacking faults of the unit layers along the *c* axis.



Fig. 10. (a,b) SAED pattern and indexing of syntaxic intergrowths between three polytypes in B<sub>2</sub>S corresponding to zone [11 20].

# 3. Conclusions

A mixed-layer member of the Ca-RE fluorocarbonate mineral series  $B_2S$ ,  $CaCe_3(CO_3)_4F_3$ , has been investigated by X-ray single-crystal diffraction analyses and HRTEM. Its crystal structure and chemical composition show that it is a mixedlayer member between bastnaesite-(Ce) and parisite-(Ce). Its structure (*dededgfg* or BBS) is constituted by ordered stacking of layers of bastnaesite-(Ce) (de or B layer) and synchysite-(Ce) (dgfg or S layer) in the ratio 2:1 along the c axis. This study reveals that  $B_m S_n$  ( $m \ge n$ ) mixed-layer members of this series are constituted by ordered stacking of unit layers of  $B_2S$  (dededgfg or BBS layer) [4] and parisite-(Ce) (*dedgfg* or BS layer) [1,3,11,12,16] in different proportions along the c axis. For example,  $B_3S_2 = dededgfgdedgfg$  (or BBSBS) = dededgfg (or BBS) + dedgfg (or BS) [3,6];  $B_7S_4 = dededgfgdededgfgdededgfgdedgfg$  (or **BBSBBSBBSBS**) = 3dededgfg (or 3BBS) + dedgfg(or BS) [7];  $B_{10}S_6 = dededgfgdededgfgdededgfgde$ dedgfgdedgfgdedgfg (or BBSBBSBBSBSBSBS)= 4dededgfg (or 4BBS) +2dedgfg (or 2BS) [7];.... An accurate knowledge of the crystal structure and superstructure of  $B_2S$  is therefore essential to characterize minerals of this goup.

Four new polytypes  $(2H_2, 4H, 12H \text{ and } 6R)$ with a grain size greater than 0.5 µm have been discovered in the 'polycrystals' of this mineral series. All except  $2H_2$  have long-period stacking structures with c = 7.536 - 22.608 nm, and always occur as polycrystals with disordered stacking sequences and microtwins along with other mixedlayer members and other mineral species of the series. The subcell heights of these polytypes are identical, whereas their supercell heights are different. The latter reveal the variations of the stacking period and arrangement of the unit layers dededgfg in the long-period stacking sequences. These new polytypes are mainly formed by periodic variations of the relative displacements of the CO<sub>3</sub> groups between the structural layers *dededgfg* in the  $B_2S$ .

The {0001} microtwin of B<sub>2</sub>S-6*R* has been observed and identified by HRTEM. Syntactic intergrowths among the three polytypes  $2H_1$ , 6R and 4H have also been discovered, revealing variations

of stacking disorder of the structural layers dededgfg and aperiodicity in the orientation arrangement of CO<sub>3</sub> groups between the layers. This provides important microscopic information about the growth environment and mechanism of formation of this mineral series.

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