Continuous Cauchy wavelet transform analyses of EXAFS spectra: A qualitative approach

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ABSTRACT

To better understand the extended X-ray absorption fine structure (EXAFS) spectroscopic information obtained for complex materials such as those encountered in Earth materials, we propose to use the Continuous Cauchy Wavelet Transform (CCWT). Thanks to this method, EXAFS spectra can be visualized in three dimensions: the wavevector (k), the interatomic distance uncorrected for phase-shifts (R'), and the CCWT modulus (corresponding to the continuous decomposition of the EXAFS amplitude terms). Consequently, more straightforward qualitative interpretations of EXAFS spectra can be performed, even when spectral artifacts are present, such as multiple-scattering features, multi-electronic excitations, or noise. More particularly, this method provides important information concerning the k range of each EXAFS contribution, such as next nearest-neighbors identification. To illustrate the potential of CCWT analyses applied to EXAFS spectra, we present experimental and theoretical spectra obtained for thorite and zircon at the Th L_{II} and Zr K edges, respectively. Then, we present CCWT analyses of EXAFS spectra collected for amorphous materials of geochemical and environmental interest, including sodium trisilicate glass and an aqueous chloride solution, at the Mo K and Au L_{III} edges, respectively. Further studies based on CCWT phase terms are underway, in order to quantitatively characterize anharmonic information from EXAFS contributions.

INTRODUCTION

Extended X-ray absorption fine structure (EXAFS) spectroscopy is a powerful tool for investigating the short and medium range environment around a selected absorbing atom in materials like minerals, glasses, and solutions (see Brown et al. 1988; Brown et al. 1995; Henderson et al. 1995 for reviews). Structural parameters (i.e., interatomic distances, coordination numbers, Debye-Waller factors, etc.) can be accurately yielded through analysis and the modeling of the EXAFS spectra. Notably, the recent development of ab initio packages, such as FEFF (Rehr et al. 1992), GNXAS (Filipponi et al. 1995; Filipponi and Di Cicco 1995), and EXCURVE (Binsted and Hasnain 1996), allows for an efficient approach to data reduction, particularly for crystalline compounds. However, interpretations concerning complex systems (i.e., aperiodic structures, adsorption mechanisms, samples under extreme conditions, etc.) are generally more difficult. In such systems, Fourier transform (FT) analyses (Lytle et al. 1975) help with visualization of the various shells of neighboring atoms surrounding the central atom. In addition, inverse FT filtering (FT-¹) is used to extract the different components to the EXAFS signal, $\chi(k)$ (i.e., the various pseudo-periodic contributions corresponding to a specific shell of neighbors around the central atom; Lytle et al. 1975). The theoretical equation of the EXAFS signal can be written as (Sayers et al. 1970; Stern et al. 1975):

in which S_i is the amplitude reduction factor for the total central atom loss. For each shell of neighboring atoms j, N_j is the number of backscattering atoms, R_j is the average distance between the central and the backscattering atoms, $|f_j(k, \pi)|$ is the effective curved-wave backscattering amplitude function, σ_j^2 is the Debye-Waller factor, λ_j is the photoelectron mean free path, and $\Sigma \phi_{ij}(k)$ is the sum of the phase-shift functions including the backscattering phase-shifts of the central and neighboring atoms, as well as some anharmonic contributions related to thermal vibrations (Stern and Heald 1983; Teo 1986; Stern et al. 1991). Also, the wavevector k is defined by the relation:

$$k = \sqrt{2m_e(E - E_0)} / h$$

in which m_e is the electron mass, h is Planck's constant, E_0 is the threshold energy of the absorption edge, and E is the absolute energy.

Based on Equation 1, the type of backscattering neighbors can be determined based on the shape of the backscattering amplitude function, $|f_j(k, \pi)|$; the atomic number being determined at Z ± 10 (see Teo 1986). However, instead of examining each FT⁻¹ independently (with all of the problems related to the possible overlap between close shells of neighbors), it would be more efficient to observe all the EXAFS contributions at once, by plotting them in reciprocal (*k*) and direct (*R*') space simultaneously (*R*' being the phase-shifts uncorrected distance).

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To achieve this, we propose to use continuous wavelet analy-

 $[\]chi(k) \approx \sum_{j} S_i(k) \frac{N_j}{kR_i^2} \Big| f_j(k,\pi) \Big| e^{-2R_j/\lambda_j(k)} e^{-2\sigma_j^2 k^2} \sin\left(2kR_j + \sum \phi_{ij}(k)\right)$ (1)

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sis (Louis et al. 1997; Torrésani 1999) to decompose the analyzed signals in (k,R') space to perform more straightforward interpretations of EXAFS spectra collected for samples of mineralogical, geochemical, and environmental interest. These previous studies showed that continuous wavelet analysis is an efficient method for analyzing "frequency-modulated" signals. The concept of wavelet analysis was first introduced by the geophysicist Jean Morlet at the beginning of the 1970s, based on early work on "time-frequency" Haar-Gabor decompositions (see Haar 1910). In petroleum exploration, wavelet analyses are widely used to study seismic signals to help discover deep oil reservoirs (Morlet and Grossmann 1984; Goupillaud et al. 1984). Since that pioneering work, other types of wavelet analyses were developed for a broad range of applications such as video and audio compression (see Daubechies 1988 among many others). Applied to EXAFS spectroscopy, wavelet theory was first used to remove the atomic background (Shao et al. 1998), but also to reconstruct the radial distribution functions (Yamaguchi et al. 1999). In this study, we propose to use the continuous Cauchy wavelet transform (CCWT; Argoul and Le 2003) to decompose the EXAFS signal in reciprocal and real space simultaneously. Then EXAFS spectra can be visualized in three dimensions (3D): the wavevector (k), the interatomic distance uncorrected for phase-shifts (R'), and the CCWT modulus. Consequently, CCWT analysis provides an informative 3D view of the (k,R') dependency of each EXAFS component; such information is particularly useful for the identification of the various contributions composing an EXAFS spectrum. In this study, we present several examples of transition elements in various geomaterials: two crystals, a silicate glass, and an aqueous solution.

CONTINUOUS CAUCHY WAVELET TRANSFORM

Numerical description

The continuous wavelet transform of a given frequencymodulated signal $\chi(k)$ is defined as follows (Chui 1992; Torrésani 1999):

$$T_{\Psi}[\chi](b,a) = \left\langle \chi(k), \Psi_{(b,a)}(k) \right\rangle = \frac{1}{a} \int_{-\infty}^{+\infty} \chi(k) \overline{\Psi}\left(\frac{k-b}{a}\right) dk$$
(2)

in which $\langle \chi(k), \psi_{(b,a)}(K) \rangle$ is the scalar-product of the two functions $\chi(k)$ and $\psi_{(b,a)}(k)$. Applied to EXAFS spectra, $\chi(k)$ represents the EXAFS signal (Eq. 1), usually k^3 -weighted and multiplied by a smooth apodisation window. The function $\psi_{(b,a)}(k)$ represents a so-called "family" of wavelets, characterized by a constant shape and variable sizes ($\overline{\psi}$ denotes the conjugate of ψ). Also, the parameters *b* and *a* are related to the *k* and *R*' spaces, respectively. The variable *b* corresponds to any value of the *k* vector, whereas the variable *a* (so-called "scaleparameter") is defined here as n/2R', in which *n* is a parameter related to the type of wavelet used (see below for details).

The numerical computation of the continuous wavelet transform is based on a fast Fourier transform algorithm (Cooley method). Then, following Argoul et al. (1998), Equation 2 can be written as:

$$T_{\Psi}[\chi](b,a) = \frac{1}{\pi} \int_{-\infty}^{+\infty} \hat{\chi}(R') \overline{\hat{\Psi}}(2aR') e^{2ibR'} dR'$$
(3)

in which *i* is the complex number. The Fourier transform $\hat{\chi}(R')$ of the signal $\chi(k)$ is defined as:

$$\hat{\chi}(R') = \int_{0}^{+\infty} \chi(k) e^{-2ikR'} dk$$
(4)

Consequently, Equation 3 shows that, for a given value for *a*, the wavelet transform $T_{\psi}[\chi](b,a)$ can be seen as an inverse Fourier transform of the function $\hat{\chi}(R')$ convoluted by the function $\hat{\psi}(2aR')$. Thus, the wavelet transform being a complex function, the amplitude and the phase terms of the analyzed signal are calculated from the real and the imaginary parts.

In Equation 3, the function ψ is called the "mother" or "analyzing" wavelet ($\hat{\psi}$ being the conjugate of the FT of ψ). In this study, we chose to use the complex-valued Cauchy wavelet of order *n* (*n* > 1). The wavelet transform is then called continuous Cauchy wavelet transform (CCWT). The Cauchy wavelet $\psi_n(k)$ and FT $\hat{\psi}_n(R')$ are, respectively, defined as (Argoul and Le 2003):

$$\Psi_{n}(k) = \left(\frac{i}{k+i}\right)^{n+1} \cdot \hat{\Psi}_{n}(R') = 2\pi \frac{R'^{n}}{n!} e^{-R'} H(R')$$
(5)

in which H(R') is the Heaviside step function. Also, the Cauchy parameter *n* controls the resolutions, Δk and $\Delta R'$, of the CCWT in the *k* and *R*' spaces, respectively. Following Argoul and Le (2003), the distribution of resolutions is defined by:

$$[k - \Delta k, k + \Delta k] \times [R' + \Delta R_1', R' + \Delta R_2']$$
(6)

with a symmetric shape in reciprocal space:

$$\Delta k = \frac{1}{R'} \left(\frac{n}{2\sqrt{2n-1}} \right) \tag{7}$$

and an asymmetric shape in direct space:

$$\Delta R_{1}' = R' \left(\frac{1}{2n} - \frac{\sqrt{2n+1}}{2n} \right) ; \ \Delta R_{2}' = R' \left(\frac{1}{2n} + \frac{\sqrt{2n+1}}{2n} \right)$$
(8)

In Equations 7 and 8, note that Δk and $\Delta R'$ are inversely proportional. Moreover, Δk and $\Delta R'$ are constrained by the Heisenberg inequality: Δk . $\Delta R' \ge 1/4$ (Chui 1992). Thus, for a given *n* value, $\Delta R'$ is small and Δk is high for low *R'* values. Reciprocally, $\Delta R'$ is high and Δk is small for high *R'* values.

The Cauchy mother wavelet was chosen because of its "progressive" properties, which means that its FT vanishes for R' < 0 (due to the Heaviside step function). An example of the application of the Cauchy wavelet for two-dimensional continuous wavelet analysis is the determination of "hidden" symmetries in the crystal structure of quasi-crystalline alloys (see Antoine et al. 1999a, 1999b). Moreover, the use of a complex-valued wavelet (rather than a real one, i.e., $\in \mathbb{R}$) simplifies the numerical computation because of its real FT, and is well suited for analyzing frequency-modulated signals (Torrésani 1999; Argoul and Le 2003).

As the asymptotic properties are respected for the analyzed signal (i.e., the phase term is varying faster than the amplitude term; Argoul et al. 1998), the CCWT modulus tends to concentrate all the information related to EXAFS contributions in (k,R')space, near a series of curves called "CCWT ridges" (see Carmona et al. 1997; Carmona et al. 1998). For that reason, the analyzed EXAFS signal is generally k^3 -weighted and multiplied by an apodisation window before computation. Then, according to Carmona et al. (1998), Equation 3 can be written as:

$$T_{\Psi_n}[\chi](b,a) \approx \frac{1}{2} \sum_{j} \hat{\Psi}_n(a.\Phi'_j(b)) \mathbf{A}_j(b) \ e^{i\Phi_j(b)}$$
(9)

in which $A_j(b)$ represents the EXAFS amplitude term and $\Phi_j(b)$ the EXAFS phase term for each j^{th} pseudo-periodic contribution related to a shell of backscattering neighbors (see Eq. 1); $\Phi'_i(b)$ being the derivative of $\Phi_i(b)$.

In the (b,a) space of the CCWT, each j^{th} ridge $a_j(b)$ is defined as $n/\Phi'_j(b)$ (Carmona et al. 1997; Carmona et al. 1998). Therefore, when the scale-parameter *a* is localized on a given j^{th} ridge [i.e., $a = a_j(b)$], the CCWT phase corresponds to $\Phi_j(b)$, and the CCWT modulus becomes:

$$\left|T_{\Psi_n}[\boldsymbol{\chi}]\right|(b,a=a_j(b)) \approx \frac{1}{2}\hat{\Psi}_n(n) \mathbf{A}_j(b)$$
(10)

Consequently, on each *j*th ridge observed, the CCWT modulus provides the amplitude term $A_j(b)$ of a given EXAFS signal contribution to within a wavelet-defined constant $(\frac{1}{2})\hat{\Psi}_n(n)$.

In this paper, each EXAFS spectrum was previously k^3 -weighted, and multiplied by the Kaiser-Bessel window with a Kaiser-Bessel parameter of 4 (Bonnin et al. 1985). Moreover, all the CCWT presented were calculated with n = 200, and typically, 400 values were used for the scale parameter *a* (corresponding to the number of pixels in the *R*' dimension). Therefore, for *R*' values around 2 Å, the resolutions in the *k* and *R*' spaces are, respectively, $\Delta k \approx \pm 2.5$ Å⁻¹ and $\Delta R' \approx \pm 0.1$ Å, and for values around 6 Å, $\Delta k \approx \pm 0.8$ Å⁻¹ and $\Delta R' \approx \pm 0.3$ Å.

Application to EXAFS spectroscopy

This study focuses mostly on qualitative analyses of EXAFS spectra, based on the study of CCWT-filtered EXAFS amplitude terms. Teo (1986) has shown that the variations in amplitude of a given normalized EXAFS spectrum are substantially affected by the backscattering amplitude functions of the neighboring atoms ($|f_i(k, \pi)|$ in Eq. 1); this being related to the number of (repulsive) electrons in the electronic cloud of the backscattering neighbors. Therefore, atoms having high atomic numbers are more efficient backscatterers at high k values, in contrast to atoms having lower atomic numbers, which are more efficient backscatterers at lower k values. Consequently, since the CCWT modulus is a decomposition of each jth EXAFS amplitude term in R' space (see Eq. 10), the interpretation of a CCWT calculation is here essentially based on the graphical analysis of its modulus. Similarly, the CCWT phase corresponds to the decomposition of each j^{th} EXAFS phase term in R' space. However, despite the fact that the CCWT phase contains valuable information, the qualitative interpretation of the neighboring atomic shells is much less direct (see Eq. 1), and its analysis will be presented in a forthcoming study.

By simply comparing the relative variations for each ridge

of the CCWT modulus, we obtain qualitative information about the various contributions to the EXAFS spectrum. To illustrate and validate the use of CCWT analysis applied to EXAFS spectra, we present a study of Th and Zr in two crystalline model compounds: thorite and zircon, respectively. Then we apply the CCWT method to the analysis of experimental EXAFS spectra collected for molybdenum and gold in two aperiodic systems: a sodium trisilicate glass and an aqueous chloride solution.

RESULTS

Crystalline thorite

A synthetic crystalline α-thorite (ThSiO₄) was investigated using EXAFS spectroscopy at the Th L_{III} edge (see Farges and Calas 1991). Figure 1 shows the EXAFS spectrum with corresponding FT and CCWT analyses. On the FT spectrum, each peak is identified according to the crystal structure refinement of Taylor and Ewing (1978). In agreement with this study, four main contributions are observed: (1) eight O atoms as closest neighbors (~2.0 Å on the FT plot, at an average Th-O distance of 2.41 Å), (2) two Si atoms as second neighbors (~2.9 Å on the FT plot; \langle Th-Si \rangle = 3.16 Å), (3) a mixed shell with four Si and four Th atoms as third neighbors (≈ 3.8 Å on the FT graph; $\langle Th-Si/Th \rangle = 3.90$ Å) and finally, (4) a last shell arising essentially from twelve Th atoms as next nearest-neighbors (~5.9 Å on the FT spectrum; $\langle Th-Th \rangle = 5.93-5.95$ Å). Each of these four contributions can be associated with a distinct ridge on the CCWT modulus, as shown in Figure 1c. On this diagram, the EXAFS amplitude terms arising from O, Si, and Th atoms show different shapes in k space. The 3D representation of the CCWT modulus (Fig. 2) provides a complementary view of the shape of each EXAFS amplitude term. Figures 1c and 2 show that the O atom first-neighbors contribute significantly



FIGURE 1. EXAFS analysis of crystalline thorite at the thorium L_{III} edge: (a) k^3 -weighted experimental EXAFS spectrum; (b) FT-magnitude; (c) CCWT modulus showing the (k,R') localization of each EXAFS contribution. The FT peaks and CCWT ridges are labeled according to the structure refinement of Taylor and Ewing (1978).



FIGURE 2. Three-dimensional view of the CCWT modulus calculated from the Th L_{III} edge EXAFS spectrum presented in Figure 1, showing different EXAFS amplitude terms (or ridges) for the different atomic shells.

for a *k* range of 2–14 Å⁻¹, whereas Si and Th next nearestneighbors are significant for *k* ranges of 7–13 Å⁻¹ and 11–16 Å⁻¹, respectively.

To further understand the CCWT analyses presented in Figures 1c and 2, ab initio EXAFS calculations of thorite at the Th L_{III} edge were carried out using the FEFF 7.02 package (Rehr et al. 1992; Ankudinov et al. 1998). The calculations were based on the crystal structure refinement of Taylor and Ewing (1978). To calculate theoretical EXAFS spectra, we used default settings for the atomic pair potential (i.e., Hedin-Lunqvist), and automatic overlapping of the muffin-tin radii (AFOLP option). Debye-Waller factors were adjusted to match the experimental spectra, and multiple-scattering (MS) paths were included in the calculation. The CCWT analyses of the experimental spectrum (Fig. 3a) and its theoretical counterpart (Fig. 3b) are in good agreement with each other. Also, each of the four EXAFS contributions (described above) can be identified, according to their EXAFS amplitude terms calculated by FEFF 7.02 (Fig. 3c). For example, the EXAFS contribution related to the eight O atoms is centered around 6 Å⁻¹ on the CCWT modulus (Fig. 3b), which is consistent with the maximum of its theoretical amplitude term (Fig. 3c). Similar agreements can be found for the more distant contributions (Si and Th next nearest-neighbors are centered around 9 and 15 Å⁻¹, respectively).

The ridge located near 1.2 Å in Figure 3a can be attributed to a spectral artifact because no atom can be located between the central absorbing thorium atoms and the closest O atoms. The origin of such a contribution is most likely related to multielectronic excitation features (see Filipponi et al. 1991). Indeed, these features generate discontinuities in the atomic background of the EXAFS signal (see Campbell et al. 2002), and therefore generate some "low frequencies" in the R' space (see Solera et al. 1995). According to Farges et al. (2000), up to four suspected transitions (located near 2.5, 4.0, 5.5, and 10 $Å^{-1}$) could be identified in the EXAFS of thorite at the Th L_{III} edge. However, because FEFF 7.02 does not calculate multielectronic excitations, these features are not observed in Figure 3b. Also, FEFF calculations suggest that MS contributions are less intense than those associated with single scattering. Consequently, their amplitudes are not clearly distinguishable on the CCWT modulus (Fig. 3b).

Thorite vs. zircon

Because thorite and zircon are isostructural $(I4_1/amd)$, it is possible to highlight the effect of the cationic substitution, Th \leftrightarrow Zr, on the CCWT modulus. The EXAFS spectrum of crystalline zircon (ZrSiO₄) at the Zr K edge was calculated using FEFF 7.02, according to the crystal structure refinement of Robinson et al. (1971). The calculation was performed using the same conditions as before. Also, the MS paths of the photoelectrons were included in the calculation. However, their amplitudes were relatively low. Therefore, the main contributions to the EXAFS spectrum essentially arise from the singlescattering paths. CCWT analyses of theoretical EXAFS spectra are shown in Figures 4a and 4b for thorite and zircon, respectively. Both images show similarities concerning, in particular, the maximums of the ridges related to O and Si atoms along the k axis. In contrast, the two ridges related to zirconium in Figure 4b $(\langle Zr-Zr \rangle = 3.63 \text{ and } 5.55 \text{ Å}; \text{ Robinson et al. 1971})$ show maximums near 11 Å⁻¹, whereas those arising from thorium (<Th-Th> = 3.90 and 5.95 Å; Taylor and Ewing 1978) in Figure 4a are localized near 13-14 Å-1. That difference can be attributed to the different atomic numbers of Th and Zr (90 and 40, respectively).



FIGURE 3. Analysis of the Th L_{III} edge EXAFS spectrum for crystalline thorite: (a) CCWT of the experimental spectrum; (b) CCWT of the theoretical spectrum (using FEFF 7.02); (c) path decomposition, based on FEFF 7.02 calculations.



FIGURE 4. Comparison of the CCWT modulus for crystalline thorite (a) and zircon (**b**): (bottom) k^3 weighted EXAFS spectra calculated with FEFF 7.02 at the Th $L_{\rm III}$ edge and the Zr K edge for thorite and zircon, respectively; (top) CCWT analyses of the EXAFS spectra. The ridges on the CCWT modulus are labeled according to the structure refinement of Taylor and Ewing (1978) and Robinson et al. (1971) for thorite and zircon, respectively.

DISCUSSION

Molybdenum in a sulfur-bearing silicate glass

A sodium trisilicate glass (Na₂Si₃O₇) doped with 2000 ppm of molybdenum was synthesized under controlled oxygen and sulfur fugacities of 10^{-10.2} and 10^{-1.6} atm, respectively (Siewert et al. in preparation). An experimental EXAFS spectrum was collected at the Mo K edge up to 14 Å⁻¹ (Fig. 5a). Its FT spectrum (Fig. 5b) shows two peaks having approximately the same height, centered around 1.4 and 1.8 Å. Those two peaks are related to a double shell of first-neighbors. However, using a simple FT analysis it is not possible to determine if they arise from two O atom shells, two S shells, or a mixture of both. Moreover, the two contributions are too close to each other ($\Delta R' \approx 0.4$ Å) to be filtered using the FT⁻¹ method, and only a double shell fit can be used to identify them. In contrast, with only one CCWT calculation, a 3D representation of the modulus (Fig. 5c) clearly shows different peak shapes for the two contributions. Therefore, two different types of atoms are surrounding the central molybdenum atom. Moreover, Figure 5c shows that the closest contribution (around 1.4 Å) is centered on relatively low k values as compared to the more distant one (around 1.8 Å). Consequently, because only two types of anions, having different atomic numbers, are present in this glass (i.e., O, Z = 8 and S, Z = 16), one can infer that the closest contribution is most likely related to O atoms and the other is most likely related to S atoms (see the previous sections for details). Those results indicate that molybdenum forms oxy-sulfide complexes in sodium trisilicate glasses, helping to understand its transport properties in magmatic systems (see Siewert et al. in preperation).

Gold in an aqueous chloride solution

An aqueous chloride solution equilibrated at pH = 9.2 and containing 0.01 M of gold was investigated using EXAFS spectroscopy at the Au L_{III} edge (Farges et al. 1993). The EXAFS

spectrum, as well as FT and CCWT analyses, are presented in Figures 6a, 6b, and 6c, respectively. Here again, a double shell of first-neighbors surrounds the absorbing atom. Two peaks are present on the FT (Fig. 6b), the closest contribution (around 1.7 Å) being approximately two times higher than the other (around 2.0 Å). The CCWT modulus (Fig. 6c) presents two different shapes for each EXAFS amplitude term, suggesting the presence of two different types of anions around the gold atom. The closest peak (≈ 1.7 Å) is centered near 6 Å⁻¹, whereas the other (≈ 2.0 Å) is centered near 8.5 Å⁻¹. As for the previous example, we can infer that the closest contribution is most likely related to O atoms because of the relatively low atomic number (Z = 8), whereas the other is most likely related to Cl atoms (Z = 17). This result is in excellent agreement with the study of Peck et al. (1991), showing the presence of Au(OH)₂Cl₂ moieties, with gold atom located in square planar polyhedra. Also, the study of Farges et al. (1993) suggests the presence of two O atoms located at 1.97 Å and two Cl atoms located at 2.28 Å from the Au central atom.

Figure 6c shows a "gray spot" localized between 3 and 4 Å in *R*' space, and centered below the oxygen one in *k* space, from around 2 to 8 Å⁻¹. However, the square-planar geometry around gold (i.e., AuX₄ polyhedra, where X = O/Cl) involves photoelectron MS paths of order 3 and 4, generating three main contributions between 3.4 and 4.0 Å (Berrodier et al. 1999). Therefore, this "gray spot," corresponding to a combination of CCWT ridges, can be reasonably related to MS contributions.

In addition, when k values increase, "high-frequencies" related to noise interfere progressively with "low-frequencies" (i.e., low R' values) arising from the different contributions to the EXAFS signal. Indeed, the EXAFS spectrum in Figure 6a shows an increasing noise level with increasing k values. At the same time, the CCWT modulus (Fig. 6c) highlights the noise domain, which is delimitated from the structural contributions



FIGURE 5. (a) k^3 -weighted EXAFS of the Na₂Si₃O₇ glass (2000 ppm of Mo and 2 wt% of S), collected at the Mo *K* edge; (b) FT analysis of the EXAFS spectrum; (c) three-dimensional view of the CCWT modulus (localized between 0 and 3 Å) calculated from the EXAFS spectrum. The 3D graph highlights the presence of a mixed environment around Mo, related to oxy-sulfide complexes.

and from the MS features (occurring below 10 Å⁻¹), as shown by the solid line. Consequently, the CCWT modulus provides an interesting representation, where structural contributions, MS features, and noise are located in different regions in (k,R')space.

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FIGURE 6. (a) k^3 -weighted EXAFS spectrum collected at the Au L_{III} edge in an Au (0.01 M)-aqueous chlorine solution (pH = 9.2); (b) FT analysis of the EXAFS spectrum; (c) CCWT analysis of the EXAFS spectrum. "O" and "CI" represent O atom and chlorine neighboring shells, respectively. "MS" corresponds to the spot related to the multiple-scattering effects arising from the first-neighbors. The solid line separates the MS features and the single-scattering contributions from the noise.

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