## CONTROLLED POLYMERIZATION OF METANILIC ANION WITHIN THE INTERLAYER OF NIAI LAYERED DOUBLE HYDROXIDE

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**Abstract**—The controlled chemical oxidative polymerization of metanilic anion  $(m\text{-}NH_2\text{C}_6\text{H}_4\text{SO}_3^-)$  within the interlayer of NiAl layered double hydroxide was performed using, for the first time, ammonium formation<br>interlayer<br>sulfonic<br>ntities of the interlayer of  $\mathbf{A}$  is a set  $\mathbf{A}$  and  $\mathbf{A}$  is a set  $\mathbf{A}$  is a set  $\mathbf{A}$  and  $\mathbf{A}$  is a set  $\mathbf{A}$  is a s polymerization was investigated systematically and it was found that interleaved polyaniline sulfonic (PANIS) was present in different oxidation states and protonation levels when different quantities of external oxidizing agents were added. A mechanism for the oxidative polymerization of metanilic anion in NiAl layered double hydroxide is proposed, based on the intercalation of the oxidizing agent and the interlayer polymerization of monomer. The resulting PANIS/NiAl LDH composites were characterized by powder X-ray diffraction, ultraviolet-visible absorption spectra, Fourier transform infrared and X-ray photoelectron spectroscopy. photoelectron spectroscopy.

Key WordsÐIn Situ Polymerization, Layered Double Hydroxide, Polyaniline.

INTRODUCTION<br>Recently, organic/inorganic nanocomposite materials have attracted extensive attention due to their unusual physicochemical properties which cannot be achieved by conventional solid-state reactions (Leroux and Besse,  $2001$ ). One focus of this investigation is the intercalation of conducting polyconjugated polymers, such as polyaniline, polythiophene and polypyrrole etc., into the <sup>4</sup>galleries' within layered double hydroxides (LDHs)<sup>\*</sup><br>which are excellent anion exchange materials and can be represented by the general formula  $[M_{1-x}^{\text{II}}M_x^{\text{III}}]$ represented by the general formula  $[M_{1-x}M_x]$ <br>(OH)<sub>2</sub>]<sup>x+</sup>( $A^{n-1}$ <sub>x/n</sub>·mH<sub>2</sub>O (where  $M^{II}$  and  $M^{III}$  are metall<br>cations, A is an anion) (Moujahid *et al.*, 2002; Vieille<br>*al.*, 2004). In the conjugated polymer/LDH nan (OH)<sub>2</sub>]  $(A^{-1})_{x/n}$   $m$ H<sub>2</sub>O (where M<sup>-2</sup> and M<sup>-22</sup> are metallic cations, A is an anion) (Moujahid *et al.*, 2002; Vieille *et al.*, 2004). In the conjugated polymer/LDH nanocomposites, the polymers act as guests or func  $al., 2004$ ). In the conjugated polymer/LDH nanocomposites, the polymers act as guests or functional materials which can be obtained by in situ polymerization of the interlayer monomer or by direct intercalation; the constrained interlamellar region of LDHs serves as a templating agent, positioning and orienting the polymer molecules within the interlayer. The interaction between hosts and guests at the nanometer scale integrates the merits of each component into the novel nanocomposites which will potentially possess electrochemical, optical or magnetic applications. Several conjugated polymer/ LDH composites have been synthesized and studied, such as polyaniline/CuCr LDH, polyaniline/CuAl LDH (Challier and Slade, 1994), polyaniline/LiAl LDH (Isupov et al., 2001), thiophene dimers/ZnAl LDH, thiophene dimers/ZnCr LDH (Tronto et al., 2004).

It is well known that the oxidative polymerization of aniline gives rise to polyaniline existing in a variety of aniline gives rise to programme existing in a variety of  $\sum_{i=1}^{n}$  to  $\sum_{i=1}^{n}$  to  $\sum_{i=1}^{n}$  to  $\sum_{i=1}^{n}$  and  $\sum_{i=1}^{n}$  as  $\sum_{i=1}^{n}$  as  $\sum_{i=1}^{n}$  as  $\sum_{i=1}^{n}$ 

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depicted in Figure 1, differing greatly in color (from colorless to violet) and electrochemical properties (from insulator to metal) (Wei et al., 1999). As a result, controlling the polymerization of the aniline intercalated in the restricted region of LDHs under given conditions and keeping the polymer interlayered at electroactive states are necessary for applications of the novel polyaniline/LDH nanocomposite materials.

In this study, the controlled chemical oxidative polymerization of metanilic anion  $(m-NH_2C_6H_4SO_3^-)$ within the interlayer region of NiAl LDH has been <sup>3</sup>)<br>en<br>on performed, by the use of ammonium persulfate as the performing by the use of anti-term performance as the use of any of NiAl LDH has been used as novel nano-reaction vessels for the *in situ* polymerization of interlayer monomers (Moujahid et al., 2003). The effect of the quantity of oxidizing agent on the oxidation states and the protonation levels of the PANIS formed in the interlayer has been investigated, and the mechanism of the interlayer oxidative polymerization has been proposed. The characteristics of the resulting PANIS/NiAl LDH composites were identified through X-ray diffraction (XRD), ultraviolet-visible spectroscopy (UV-vis), Fourier transform infrared spectroscopy (FTIR) and X-ray photoelectron spectroscopy (XPS). Therefore, the polymerization of metanilic anion in the restricted environment between LDH sheets creates new opportunities for further research into chemical reactions in inorganic hybrids at the nanoscale and for their use in incregative hybrids at the nanoscale and for their use in the<br>catalytic andications  $\mathbf{r}$  and  $\mathbf{r}$  applies applications.

### **EXPERIMENTAL**

Reagents<br>All the chemicals used in this synthesis, including All the chemicals used in this synthesis, including  $N_{\rm e}$  (NO<sub>3</sub>)2.6H<sub>2</sub>O, Al(NO<sub>3</sub>)3.9H<sub>2</sub>O, NaOH, metalilic acid



Figure 1. Variable redox states and doping levels of polyaniline.

 $(m\text{-}NH_2C_6H_4SO_3H)$ , and  $(NH_4)_2S_2O_8$  were of analytical grade, purchased from Aldrich, and used without further purification. All solutions were prepared using distilled and decarbonated water.

*Synthesis*<br>The precursor LDH  $[Ni_2Al(OH)_6](NO_3) \cdot nH_2O$  was prepared following a standard aqueous co-precipitation and thermal crystallization method (Wilson et al., 1999). A solution of NaOH (8.0 g, 0.20 mol) in water  $(100 \text{ cm}^3)$  was added dropwise over 2 h to 100 cm<sup>3</sup> of (100 cm<sup>-</sup>) was added dropwise over 2 h to 100 cm<sup>-</sup> of<br>a solution containing Ni(NO, )...6H, O (10.4 g 0.066 mol) and  $Al(NO<sub>3</sub>)<sub>3</sub>·9H<sub>2</sub>O$  (12.5 g, 0.033 mol) with vigorous stirring under a nitrogen atmosphere. The value of the solution pH at the end of the addition was 6.0. The resulting gelatinous precipitate was maintained at 100°C for 72 h, centrifuged and then washed thoroughly until the pH of the supernatant solution was 7.0. The powder was dried at  $70^{\circ}$ C for 24 h under vacuum.

The metanilic intercalated NiAl LDH was prepared following the ion-exchange method. A 100  $\text{cm}^3$  solution of metanilic acid  $(5.2 \text{ g}, 3.0 \text{ mmol})$  with pH adjusted to 6.5 was added to a suspension of NiAl-NO<sub>3</sub> LDH  $(5.0 \text{ g},$ ~1.5 mmol) in water (100 cm<sup>3</sup>). The mixture was held at  $\frac{30\%}{100}$  under a nitrogen atmosphere for 72 h. The product  $30^{\circ}$ C under a nitrogen atmosphere for 72 h. The product was washed extensively with water, centrifuged and dried under vacuum for 24 h.

The PANIS/LDH was synthesized by oxidizing the metanilic LDH through an external oxidizing agent  $\left[\frac{(NH_4)_2S_2O_8\right]$ . Three samples were prepared with different quantities of the oxidizing agent in order to investigate its effect on the reaction products. Powders of  $(NH_4)_2S_2O_8$  weighing 0.002 g, 0.02 g and 0.2 g were added to a suspension of  $2.0 \text{ g}$  metanilic LDH in 100 cm<sup>3</sup> water under a nitrogen atmosphere. The pH of<br>the slurries was adjusted to 6.5 using NaOH (2.0 M) and the slurries was adjusted to 6.5 using NaOH (2.0 M) and maintained at room temperature for 72 h. The products were washed extensively, centrifuged and dried under vacuum for 24 h.

### Characterization Characterization

Powder XRD measurements were performed on a Rigaku XRD-6000 diffractometer, using  $CuK\alpha$  radiation  $(\lambda = 0.154$  nm) at 40 kV, 30 mA. Data were collected over the angular range from 3 to  $70^{\circ}2\theta$  at a scanning rate of  $0.02^{\circ}$  s<sup>-1</sup> at room temperature.

of 0.02° s  $\cdot$  at room temperature.<br>The UV-vis spectra were comode using a Shimadzu UV-2401<br>the  $\sim$ 240–800 nm region with the The UV-vis spectra were contributed in reflections. the  $\approx$  240–800 nm region with the samples dissolved in 0.1 M hydrochloric acid solution.<br>Fourier transform infrared spectra were recorded

0.1 M hydrochloric acid solution.<br>Fourier transform infrared spectra were recorded<br>using a Vector22 (Bruker) spectrophotometer in the<br>range 4000 to 400 cm<sup>-1</sup> with  $2 \text{ cm}^{-1}$  resolution. A<br>standard KBr disk (1 mg of samp using a Vector22 (Bruker) spectrophotometer in the range  $4000$  to  $400 \text{ cm}^{-1}$  with  $2 \text{ cm}^{-1}$  resolution  $\Delta$ range 4000 to 400 cm with 2 cm Fesolution. A<br>standard KBr disk (1 mg of sample in 100 mg of KBr)<br>was used.<br>The chemical compositions, the local environmental  $s = 100$  mg of  $s = 100$  mg was used.<br>The chemical compositions, the local environmental

changes of the atoms and the quantitative evaluation of the various oxidation states of PANIS were determined by X-ray photoelectron spectroscopy (XPS) using a VG Scientific ESCALab220i-XL (VG Scientific Ltd., UK) spectrometer. Before scanning, the sample was sputtered using Ar gas to remove absorbed impurities on the surface. In recording the spectra,  $AIK\alpha$  irradiation was employed as the photon source with a primary tension of  $12$  kV and an emission current of 20 mA. The pressure of the analysis chamber during the scans was  $\sim$ 3 × 10<sup>-9</sup> mbar. In the data analysis, the binding energy (BE) of the core level C  $1s$  peak was set at 284.5 eV to compensate for surface-charging effects. The satellite peaks were removed for all element peaks before curve fitting. The iteration curve fit program (Levenberg-Marquardt method) was customized in order to meet the specific needs of resolving  $N$  1s peaks in the spectrum. The experimental spectra were fit with Gaussian profiles. The surface elemental compositions were determined by the ratios of peak areas corrected with empirical sensitivity factors.

Elemental analysis was performed with a Shimadzu ICPS-7500 instrument. The C, H and N contents were ICPS-7500 instruments were content to content were  $\sigma$  and  $\sigma$  and  $\sigma$ 

| Samples        | Chemical composition   |  |  |  |
|----------------|--|--|--|--|
| $NiAl-NO3 LDH$ | $\mathrm{Ni}_{0.67}\mathrm{Al}_{0.33}(\mathrm{OH})_{2}(\mathrm{NO}_{3})_{0.33}\cdot0.74\mathrm{H}_{2}\mathrm{O}$ |  |  |  |
| Metanilic-LDH  | $\rm Ni_{0.66}Al_{0.34}(OH)_{2}(C_{6}H_{4}NH_{2}SO_{3})_{0.23}(NO_{3})_{0.11} \cdot 0.78H_{2}O$                  |  |  |  |
| Sample 1       | $\rm Ni_{0.67}Al_{0.33}(OH)_{2}(C_{6}H_{4}NH_{2}SO_{3})_{0.23}(NO_{3})_{0.08}(SO_{4})_{0.01} \cdot 0.78H_{2}O$   |  |  |  |
| Sample 2       | $\rm Ni_{0.67}Al_{0.33}(OH)_2(C_6H_4NH_2SO_3)_{0.23}(NO_3)_{0.02}(SO_4)_{0.04} \cdot 0.77H_2O$                   |  |  |  |
| Sample 3       | $Ni_{0.66}Al_{0.34}(OH)_{2}(C_{6}H_{4}NH_{2}SO_{3})_{0.22}(SO_{4})_{0.06} \cdot 0.74H_{2}O$                      |  |  |  |

Samples 1, 2 and 3 refer to NiAl-metanilic LDH treated with 0.002 g, 0.02 g and 0.2 g (NH<sub>4</sub>)<sub>2</sub>S<sub>2</sub>O<sub>8</sub>, respectively.

instrument. The chemical compositions of  $NiAl-NO<sub>3</sub>$ <br>LDH and the organic intercalated LDHs are reported in LDH and the organic intercalated LDHs are reported in<br>Table 1 Table 1.

# RESULTS AND DISCUSSION

Powder X-ray diffraction<br>The powder XRD patterns of the NiAl-NO<sub>3</sub> LDH, NiAl-metanilic LDH and the oxidative products are shown in Figure 2. Table 2 lists the corresponding basal spacing and lattice parameters. The reflections of all the materials can be indexed to a hexagonal lattice with  $\bar{R}3m$ rhombohedral symmetry, commonly used for the description of the LDH structures (Whilton et al., 1997). The XRD pattern for NiAl-metanilic LDH exhibits the characteristic reflections for inorganic LDH materials. The layer thickness for the new compound is  $\sim$ 1.63 nm, which is 0.74 nm greater than that of the NiAl-NO<sub>3</sub> LDH. The expansion of the basal spacing resulted from the intercalation of metanilic anion into the LDH interlayer. Thus, the interaction of  $NiAl-NO<sub>3</sub>$  LDH with an aqueous solution of metanilic acid at pH 6.50 led to the anion exchange of  $NO_3^-$  for the acid at pH 6.50 led to the anion exchange of  $NO_3$  for the<br>anions of metanilic acid with the conservation of the<br>layered structure. anisons of metanizies actid with the conservations of the<br>layarad structure layered structure.



Figure 2. XRD patterns for NiAl-NO<sub>3</sub> LDH (a), NiAl-metanilic LDH (b), and NiAl-metanilic LDH treated with 0.002  $g(c)$ ,  $\overline{0.02 \text{ g}}$  (d) and  $\overline{0.2 \text{ g}}$  (e) of (NH $\overline{0.850z}$  $0.02525$ 

An interlayer thickness of 1.15 nm for NiAl-metanilic<br>LDH was obtained by subtracting the thickness of the Al hydroxide sheet  $(0.48 \text{ nm})$  from the layer thickness  $(1.63 \text{ nm})$ . This thickness is greater than the length of a single metanilic anion chain oriented normal to the hydroxide sheets (0.64 nm, calculated using Chemwin  $(6.0)$ , but it is less than twice that value. This may indicate that the monomer anions adopted an interpenetrating arrangement leaving interstices between the monomer anions and the hydroxide sheet. Water molecules and nitrate anions may occupy these interstices. The structure of LDHs is based on brucite-like sheets, where octahedrally coordinated metal ions share edges to form infinite sheets (Brindley and Kikkawa, 1979). The area within the  $00l$  plane occupied by each  $[Ni_{1-x}Al_x(OH)_2]^{x+}$  octahedral unit is related to the unit-[N<sub>1-x</sub>Al<sub>x</sub>(OH)<sub>2</sub>] octahedral unit is related to the unit-<br>cell parameter *a* by the formula:  $S = \sqrt{3(a^2/2)}$  (Yun and<br>Pinnavaia, 1995). The value of *a* can be calculated from<br> $d_{110}$  as determined using the XRD patter cell parameter a by the formula:  $S = \sqrt{3(a^2 + b^2)}$ <br>Pinnavaia, 1995). The value of a can be ca Pinnavaia, 1995). The value of  $a$  can be calculated from  $d_{110}$  as determined using the XRD pattern in Figure 2b  $a = 2d_{110} = 0.30$  nm). The area of each octahedral unit is therefore  $0.78 \text{ nm}^2$ , giving one positive charge per is therefore  $0.78$  nm<sup>2</sup>, giving one positive charge per dimension of the anion was calculated as  $0.60 - 0.70$  nm, and thus the corresponding cross-sectional area of the complex anions can be estimated to be in the range  $2.80 - 3.80$  nm<sup>2</sup>. Consequently, it is in the range 2.80–5.80 nm<br>impossible for the metanilic ani<br>positive charge of the hydroxic<br>co-intercalation of NO<sub>3</sub>. The<br>water was also supported by t<br>analysis (Table 1). Based on th impossible for the metanilic anion alone to balance the positive charge of the hydroxide sheet, suggesting the co-intercalation of  $NO<sub>3</sub><sup>-</sup>$ . The presence of nitrate and water was also supported by the results of elemental<br>analysis (Table 1). Based on the discussion above, the<br>structure of NiAl-metanilic LDH can be represented as analysis (Table 1). Based on the discussion above, the structure of NiAl-metanilic LDH can be represented as alternating NiAl-hydroxide sheets and interlayers containing metanilic anions, nitrate anions and water molecules. The schematic model of NiAl-metanilic LDH is illustrated in Figure 3.

The XRD patterns of the samples to which 0.002 g, 0.02 g and 0.2 g of  $(NH_4)_2S_2O_8$  were added, are shown in Figure 2c $-e$ , respectively. Apparently, the values of lattice parameter *a* for the three products are the same.<br>However, the  $d_{003}$  values of the three products decreased<br>slightly from 1.61 nm (0.002 g (NH<sub>4</sub>)<sub>2</sub>S<sub>2</sub>O<sub>8</sub>) to 1.51 nm<br>(0.2 g (NH<sub>4</sub>)<sub>2</sub>S<sub>2</sub>O<sub>8</sub>), indicating lattice parameter a for the three products are three moducts. slightly from 1.61 nm (0.002 g (NH<sub>4</sub>)<sub>2</sub>S<sub>2</sub>O<sub>8</sub>) to 1.51 nm  $(0.2 \text{ g } (\text{NH}_4)$ <sub>2</sub> $\text{S}_2\text{O}_8)$ , indicating that a reorientation of the interlamellar species occurred. This phenomenon is intermination species continued. This phenomenon is  $\mathbf{r}$  of an interved in the case of an in situ polymerization,

| <b>LDHs</b>           | $NiAl-NO3- LDH Metanilic-LDH$ |      | Sample 1 | Sample 2 | Sample 3 |
|-----------------------|-------------------------------|------|----------|----------|----------|
| $d_{003}$             | 0.89                          | 1.63 | 1.61     | 1.58     | 1.51     |
| $d_{110}$             | 0.15                          | 0.15 | 0.15     | 0.15     | 0.15     |
| interlayer spacing    | 0.41                          | 1.15 | 1.13     | 1.10     | 1.03     |
| lattice parameter a   | 0.30                          | 0.30 | 0.30     | 0.30     | 0.30     |
| lattice parameter $c$ | 2.67                          | 4.89 | 4.83     | 4.74     | 4.53     |

Table 2. Lattice parameters (nm) of NiAl-No3 LDH and the intercalated materials.

Samples 1, 2 and 3 refer to NiAl-metanilic LDH treated with 0.002 g, 0.02 g and 0.2 g of  $(NH_4)_2S_2O_8$ , respectively.

as is illustrated by the *in situ* polymerization of  $\alpha$ , $\beta$ -aspartate into hydrotalcite-type compounds (Whilton *et*  $al.$ , 1997), vinyl and amino-benzene sulfonates (VBS and ABS) into the sheets of  $Zn<sub>2</sub>A1$  LDHs (Moujahid et al., 2003), and Cu<sub>2</sub>Cr LDHs (Moujahid et al., 2005) after a soft thermal treatment ( $T = 473$  K). Therefore, the occurrence of in situ polymerization of the intercalated metanilic anion monomer in the restricted region of NiAl LDH can be deduced from the decrease in the  $c$ parameter. This was further confirmed by UV-vis, parameter. This was further confirmed by UV-vis,<br>FTIR and YPS

UV-vis spectroscopy<br>Results obtained by UV-vis spectroscopy yield strong evidence for the polymerization of the metanilic anion intercalated with NiAl LDH and provide information regarding the oxidation states of the interlayer products. Figure 4 shows the optical absorption spectra of metanilic anion and the intercalated materials in the wavelength range  $\approx$  240–800 nm. There are only two absorption bands at  $262 \text{ nm}$  and  $269 \text{ nm}$  in spectrum a.<br>These are associated with the phenyl group of the<br>metanilic monomer. No shift or intensity change of<br>these two absorption bands was observed after the<br>intercala These are associated with the phenyl group of the metanilic monomer. No shift or intensity change of these two absorption bands was observed after the intercalation of the metanilic anion into the NiAl LDH in spectrum b. Following the addition of  $(NH_4)_2S_2O_8$ , a  $\frac{1}{2}$  spectrum b. Following the addition of  $\frac{1}{2}$  (nH4)2S2 $\frac{1}{2}$  s<sub>8</sub>, and  $\mathbf{F}$  absorption band (centered at 300 nm) corresponding to the  $\pi_{\text{B}}-\pi_{\text{B}}*$  transition appeared in spectrum c, indicating the occurrence of polymerization of the intercalated monomer; the polymerization occurred between the benzenoid units. The new absorption band centered at 550 nm in spectrum d can be attributed to charge transfer from the quinoid to the benzoid segments  $(\pi_{\rm O} - \pi_{\rm B}^*)$  (Kang *et al.*, 1998), which shifted hypsochromically to 510 nm in spectrum e. Moreover, the highly<br>oxidized pernigraniline transmits at ~400 nm, which can<br>also be observed in spectrum e. The  $\pi_{Q} - \pi_{B}^{*}$  transition<br>demonstrates that quinonoid units formed in the mically to 510 nm in spectrum e. Moreover, the highly oxidized pernigraniline transmits at  $\sim$ 400 nm, which can  $\frac{1}{2}$  also be observed in spectrum e. The  $\pi_2 - \pi_2$  \* transition demonstrates that quinonoid units formed in the polymeric products (0.02 g and 0.2 g (NH<sub>4</sub>)<sub>2</sub>S<sub>2</sub>O<sub>8</sub>), and the hypsochromic shift of this absorption band indicates that the oxidation state of the interlayer product was promoted remarkably. This result is quite consistent with the report on the study of the interconversion of polyaniline oxidation states by optical absorption spectroscopy (Albuquerque et al., 2004). A similar optical spectrum has been observed for polyaniline films deposited from the vapor phase onto silica substrates, in which the optical spectrum was attributed to polyaniline units with chain length less than in 'conventional' polyaniline but<br>greater than in five ring oligomers (I use and Thomson greater than in five-ring oligomers (Luca and Thomson, 2000). Based on the comparison between our results and those in the literature, this optical spectrum of PANIS/ LDH is therefore indicative of short chains of interlayer LDH is therefore indicately to short chains to interesty to<br>PANIS in this work



Figure 3. A schematic representation of the possible arrangement for NiAl-metanilic LDH.



Figure 4. UV-vis spectra of metanilic acid (a), NiAl-metanilic LDH (b), and NiAl-metanilic LDH treated with 0.002 g (c),  $\overline{0.02 \text{ g (d)}}$  and  $\overline{0.2 \text{ g (e)}}$  of (NH  $\overline{0.8}$ ,  $\overline{0.2}$ )  $0.02528$  g (d) and  $0.25288$ .

### FTIR spectroscopy

Figure 5 shows the FTIR spectra of NiAl-NO<sub>3</sub> LDH,<br>NiAl-metanilic LDH and the three polymerization products. The broad bands around  $3448 \text{ cm}^{-1}$ (Figure 5a) and 3442 cm<sup>-1</sup> (Figure 5b) correspond to the O-H-stretching vibration of surface and interlayer water in LDH. These vibrations occur at lower frequency (Figure 5a) and 3442 cm (Figure 5b) correspond to<br>the O-H-stretching vibration of surface and interlayer<br>water in LDH. These vibrations occur at lower frequency<br>than for the O-H in free water at 3600 cm<sup>-1</sup> (Li *et al.*, water in LDH. These vibrations occur at lower frequency<br>than for the O-H in free water at 3600 cm<sup>-1</sup> (Li *et al.*,<br>2004). This suggests the formation of hydrogen bonding<br>between interlayer water and the guest anions as w where in  $\Xi$  is the continuous occur in contract  $\frac{1}{2}$  (Li et al. than for the O–H in free water at 3600 cm<sup>-</sup> (Li *et al.*, 2004). This suggests the formation of hydrogen bonding between interlayer water and the guest anions as well as the layer hydroxide groups. The absorption band at 2004). This suggests the formation of hydrogen bonding between interlayer water and the guest and the model of 1384  $cm^{-1}$  of the metanilic intercalated LDH (Figure 5b) is due to the stretching vibration of  $NO_3^-$ , and compared with the  $NiAl-NO<sub>3</sub>$  LDH precursor <sub>3</sub>,<br>or<br>ed (Figure 5a); the relative intensity of this band decreased  $\left($ - $\sigma$ <sup>-</sup> $\sigma$ <sup>--</sup>  $\sigma$ <sup>--</sup>  $\sigma$ <sup>-</sup>); the relative  $\sigma$  -metric decreased decreased decreased

remarkably. Three strong bands at 1599, 1484, 1453 cm<sup>-1</sup> and a weak one at 1315 cm<sup>-1</sup> were observed. 1453 cm  $\pi$  and a weak one at 1315 cm  $\pi$  were observed,<br>all of which can be attributed to characteristic absorp-<br>tions of metanilic anion (Liu *et al.*, 2004). Moreover, the<br>broad band at 1170 cm<sup>-1</sup> was attributed to  $\frac{1}{2}$  figure of metanilic apion (Liu *et al.* 2004). Moreover, the broad band at  $1170 \text{ cm}^{-1}$  was attributed to the presence of para-substituted aromatic C-H ( $\delta$  in plane) and the aromatic C-N (v). The bands at 1110 and 1035 cm<sup>-1</sup> aromatic C $-N$  (v). The bands at 1110 and 1035 cm<sup>-1</sup><br>were characteristic absorptions of the symmetric vibra-<br>tion of S $-C$  and S=O, respectively (Liu and Kanatzidis,<br>1995). The absorption bands of lattice vibrations<br>appear tion of  $S-C$  and  $S=O$ , respectively (Liu and Kanatzidis, 1995). The absorption bands of lattice vibrations appeared in the range  $400-650 \text{ cm}^{-1}$ .

1995). The absorption bands of lattice vibrations<br>appeared in the range  $400-650 \text{ cm}^{-1}$ .<br>The FTIR spectra of the samples treated with<br>(NH<sub>4)2</sub>S<sub>2</sub>O<sub>8</sub> reveal the different oxidation states of the<br>interlayer polymer. The appeared in the range 400–650 cm<br>The FTIR spectra of the same<br>(NH<sub>4</sub>)<sub>2</sub>S<sub>2</sub>O<sub>8</sub> reveal the different oxid<br>interlayer polymer. The intensity of th<br>at 1599, 1484 and 1454 cm<sup>-1</sup> assigne<br>of  $-B-$  and  $-Q-$  became gradu The FTIR spectra of the samples treated with  $(NH_4)_2S_2O_8$  reveal the different oxidation states of the interlay be programmed that intensity of the absorption bands of  $1500 - 1484$  and  $1454 \text{ cm}^{-1}$  assigned to the vibration of  $-B-$  and  $-Q-$  became gradually weaker with increased  $(NH_4)_2S_2O_8$  concentration (Figure 5). This reased (NH<sub>4</sub>)<sub>2</sub>S<sub>2</sub>O<sub>8</sub> concentration (Figure 5). This cates that the degree of oxidation of the polymeriza-<br>products increases with the quantity of oxidizing nt. The bands at 1523, 1474 and 1431 cm<sup>-1</sup> of the ple with indicates that the degree of oxidation of the polymerizaindicates that the degree of continuous or the programme. agent. The bands at 1523, 1474 and 1431  $cm^{-1}$  of the sample with 0.2 g of  $(NH<sub>4</sub>)<sub>2</sub>S<sub>2</sub>O<sub>8</sub>$  (Figure 5e) indicate that the polymerization product has almost a  $-Q$ structure (Huber, 2003). Comparison of IR spectra bectra<br> $\begin{bmatrix} \text{and} \\ \text{in} \\ \text{cm}^{-1} \end{bmatrix}$ ,<br>ber of between the original metanilic-intercalated LDH and its oxidation products reveals a sharp decrease in the bands at 1170 ( $v_{C-H}$  in plane), 1270 and 1315 cm<sup>-1</sup>. bands at 1170 ( $0c_{HH}$  in plane), 1270 and 1315 cm,<br>which point to a substantial decrease in the number of<br>amino groups  $(-NH_2)$ . This further confirmed the<br>formation of an interlayer polyconjugated system. amino groups  $(-NH<sub>2</sub>)$ . This further confirmed the formation of an interlayer polyconjugated system.

Formation of an interlayer polyconjugated system.<br>
Formation of an interlayer polyconjugated system.<br>
The XPS N 1s core level spectra of NiAl-metanilic<br>
LDH samples that were untreated and treated with XPS determination LDH samples that were untreated and treated with  $(NH)$ ,  $S_2O_n$  are shown in Figure 6. The main banding  $\frac{1}{2}$ s of the shown in Figure 6. The main banding



Figure 5. FTIR spectra for NiAl-NO<sub>3</sub> LDH (a), NiAl-metanilic LDH (b), and NiAl-metanilic LDH treated with 0.002 g (c), 0.02 g (d) and 0.2 g (e) of  $(NH_4)_2S_2O_8$ .



Figure 6. XPS N 1s spectra and peak-curve fit for NiAl-metanilic LDH (a), and NiAl-metanilic LDH treated with 0.002 g (b), 0.02 g (c) and 0.2 g (d) of  $(NH_4)_2S_2O_8$ .

energy moved from 399.3 eV in NiAl-metanilic LDH<br>(Figure 6a) to 399.7 eV after treatment with 0.2 g of  $(NH_4)_2S_2O_8$  (Figure 6d) and the FWHM (full width at half maximum) increased from  $1.84$  to  $2.35$  eV. It is reasonable to deduce that the chemical environment of N in the metanilic monomer has changed remarkably due to the addition of  $(NH_4)_2S_2O_8$ , and interlayer PANIS with different oxidation states formed after treatment with different quantities of  $(NH_4)_2S_2O_8$ .

The aniline polymers have the general formula  $[(-B-NH-B-NH-)_v (-B-N=Q=N-)_{1-v}]_x$  (Stejskal *et al.*, 1996), in which *B* and *Q* denote the  $C_6H_4$  rings in the benzenoid and quinonoid forms, respectively. Thus, the neutral instrinsic redox states can vary from that of the fully oxidized pernigraniline (PNA,  $y = 0$ ), to that of the fully reduced leucoemeraldine (LM,  $y = 1$ ). The 50% intrinsically oxidized polymer has been termed emeraldine (EM,  $y = 0.5$ ). In particular, the proportions of quinonoid imine  $(=N-)$ , benzenoid amine  $(-NH-)$ , and positively charged nitrogens corresponding to a<br>particular oxidation and protonation level of the polymer<br>Table 3. XPS semi-quantitative analysis of NiAl-meta and positively charged nitrogens corresponding to a  $p$  and  $p$  original oxidation and protonation level  $p$  the polymeral order  $p$ 

can be quantitatively differentiated in the properly curve-fitted  $N$  1s core level spectrum (Chen and Wen, curve-fitted N 1s core level spectrum (Chen and Wen, 2003).<br>The XPS N 1s core level spectra of the oxidized

samples (Figure  $6b-d$ ) show four contributions which can be assigned to C=N (398.3 eV), C-N (399.3 eV), C=N<sup>+</sup> (400.6 eV) and C-N<sup>+</sup> (402.8 eV) based on the binding energy of core electrons, respectively (Kang *et al.*, 1998). The relative area ratios of the four components C=N<sup>+</sup> (400.6 eV) and C-N<sup>+</sup> (402.8 eV) based on the binding energy of core electrons, respectively (Kang *et al.*, 1998). The relative area ratios of the four components as well as their respective binding energy are giv binding energy of core electrons, respectively (Kang et al., 1998). The relative area ratios of the four  $\frac{1}{2}$  components as well as their respective binding energy are given in Table 3. According to the results of previous XPS studies on chemically and electrochemically synthesized polyaniline, the oxidation level of the three products can be estimated by calculating the ratio of quinonoid imino  $(=N-)$  total peak area to benzenoid amino (-NH-) total peak area and the protonic ratio can<br>be obtained by dividing the total peak area of C-N by<br>the total peak area of C-N<sup>+</sup> (Nascimento *et al.*, 2004). It<br>can be seen from Table 3 that the samples treated be obtained by dividing the total peak area of C-N by<br>the total peak area of C-N<sup>+</sup> (Nascimento *et al.*, 2004). It<br>can be seen from Table 3 that the samples treated with<br>different amounts of  $(NH_4)_2S_2O_8$  are in differe the total peak area of  $C-N^+$  (Nascimento *et al.*, 2004). It<br>can be seen from Table 3 that the samples treated with<br>different amounts of  $(NH_4)_2S_2O_8$  are in different<br>oxidation states and protonation levels. Increasing the total peak area of C $-N$  (Nascimento *et al.*, 2004). It<br>can be seen from Table 3 that the samples treated with<br>different amounts of  $(NH_4)_2S_2O_8$  are in different<br>oxidation states and protonation levels. Increasing can be seen from Table 3 that the samples treated with different amounts of  $(NH_4)_2S_2O_8$  are in different  $\alpha$  different amounts of  $(1+\frac{1}{2})$   $\frac{1}{2}$   $\frac{1}{2}$  are in different  $\overline{\phantom{a}}$ 

Table 3. XPS semi-quantitative analysis of NiAl-metanilic LDH treated with different amounts of  $(NH_4)_2S_2O_8$ .

|          | $\equiv$ Relative peak area $(\% ) \equiv$ |                 |                 |         |   |       |       |           |         |  |  |
|----------|--|-----------------|-----------------|---------|---|-------|-------|-----------|---------|--|--|
|          |  |                 | $- (At.%) -$    | $C-N^+$ | $C=N^+$   | $C-N$ | $C=N$ | $=N-/-N-$ | $N^+/N$ |  |  |
| Samples  | Color                                      | N <sub>1s</sub> | S <sub>2p</sub> |         | $(402.8 \text{ ev})$ $(400.6 \text{ ev})$ $(399.3 \text{ ev})$ $(398.3 \text{ ev})$ |       |       |           |         |  |  |
| Sample 1 | green                                      | 3.0             | 3.4             |         |   |       |       | 0.32      | 0.14    |  |  |
| Sample 2 | blue                                       | 2.2             | 4.0             |         | 35  | 49    | 14    | 0.96      | 0.59    |  |  |
| Sample 3 | violet                                     | 2.1             | 5.3             |         | 45  | 38    | 16    | 1.56      | 0.85    |  |  |

Samples 1, 2 and 3 refer to NiAl-metanilic LDH treated with 0.002 g, 0.02 g and 0.2 g (NH<sub>4</sub>)<sub>2</sub>S<sub>2</sub>O<sub>8</sub>, respectively.

treatment with  $(NH_4)_2S_2O_8$  resulted in increased ratios<br>of  $\lceil N-1/[-NH-]\rceil$  and  $\lceil N^+ \rceil / \lceil N \rceil$  from 0.32 to 1.56 and or  $\left[\frac{m-j}{1-NH}\right]$  and  $\left[\frac{N}{N}\right]$  from 0.52 to 1.56 and 0.14 to 0.85, respectively. Thus, the quantity of oxidizing agent had a significant effect on the state of the polymerization products in this work. Furthermore,  $0.14$  to  $0.85$ , respectively. Thus, the quantity of oxidizing agent had a significant effect on the state of the polymerization products in this work. Furthermore, the colors of the 0.002 g, 0.02 g and 0.2 g  $(NH_4)_2S_2O_8$ tively, consistent with a change in their oxidation states.

As shown in Figure 6a, the peak at 406.8 eV that is generally attributed to the nitrogen of nitrate was rather strong, indicating the coexistence of  $NO<sub>3</sub><sup>-</sup>$  and the m-NH<sub>2</sub> $C_6H_4SO_3^-$  monomer between the host layers. <sup>3</sup> and the<br>ost layers.<br>along with<br>It became However, this peak decreased significantly along with whole measurement in the host layers.<br>
peak decreased significantly along with<br>
of oxidative polymerization. It became<br>
reased oxidation, and almost disappeared  $H_1$  is peak decreased significantly along with weaker with increased oxidation, and almost disappeared following the 0.2 g  $(NH_4)_2S_2O_8$  treatment (Figure 6d). This agrees with the results of the elemental analysis (Table 1), further confirming that the interlayer coexisting  $NO<sub>3</sub><sup>-</sup>$  was exchanged by the persulfate anions gradually during a process of interlayer polymerization. <sup>3</sup> was exchanged by the persulfate anions<br>ring a process of interlayer polymerization.<br>e atomic concentrations calculated from<br>g photoelectron peak areas are presented The relative atomic concentrations calculated from  $\frac{1}{2}$  relative atomic corresponding photoelectron neak areas are presented in Table 3. It can be seen that the percentage of N content decreased from  $3.4$  to  $2.1\%$  along with the increase in the percentage of S from  $3.2$  to  $5.3\%$ . A deduction can be obtained from the data listed above that  $S_2O_8^{2-}$  anions exchanged the co-existing nitrate anions in LDH and initiated the polymerization of the interlayer  $\overline{s}$  anions exchanged the co-existing nitrate anions in<br>
1 and initiated the polymerization of the interlayer<br>
omer. This might be due to the greater affinity of<br>  $\overline{s}^2$  for LDH layers than that of NO<sub>3</sub>. Additional monomer. This might be due to the greater affinity of  $\frac{1}{2}$  for IDH layers than that of NO. Additionally  $S_2O_8$  for LDH layers than that of NO<sub>3</sub>. Additionally,<br>the absorption band at 1384 cm<sup>-1</sup> assigned to the<br>presence of nitrate species decreased along with the<br>polymerization reaction (Figure 5b-e), providing the absorption band at 1384 cm<sup>-</sup> assigned to the<br>presence of nitrate species decreased along with the<br>polymerization reaction (Figure 5b-e), providing<br>further evidence for this conclusion. presence of minimal presence along with the theory

polymerization.<br>
Further evidence for this conclusion.<br> **Possible mechanism of polymerization**. Based on the<br>
experimental data above, the oxidative polymerization<br>
of the metanilic anion between the sheets of NiAl LDH Possible mechanism of polymerization. Based on the asset of the metanilic anion between the sheets of NiAl LDH can be interpreted as follows: The  $m-NH_2C_6H_4SO_3^$ anions intercalated in NiAl LDH are in an interpenetrat-3<br>t-g<br>S. ing arrangement. Water molecules and co-existing nitrate anions occupy the interlayer interstices. Following the addition of an oxidizing agent,  $S_2O_8^{2-}$ Following the addition of an oxidizing agent,  $S_2O_8^3$ <br>intercalated into the host layer by ion-exchange with<br>NO<sub>3</sub> and initiated the oxidative polymerization of the intercalated into the second into the second intercalated into the host layer second intervals.

interlayer monomer. According to the much earlier work of Mohilner *et al.* (1962), the aniline oxidation products obtained in neutral or acid media have a predominantly 'head to tail' arrangement. The oxidation state of the pa NIS formed in the interlaver was enhanced along with PANIS formed in the interlayer was enhanced along with an increase of the oxidizing agent, and the change in color of the three polymeric products confirms the corresponding promotion in the oxidation states. In summary, the interlayer oxidative polymerization of metanilic anion monomer is shown in Figure 7. metanilic anion monomer is shown in Figure 7.

## concertains

The controlled chemical oxidative polymerization of a metanilic anion monomer within the interlayer of NiAl LDH was performed using, for the first time, ammonium persulfate as the oxidizing agent. Powder XRD was used to determine the structural change of NiAl-NO<sub>3</sub> LDH and the intercalated materials. The expansion of the basal spacing gave evidence that metanilic anions were intercalated into the LDH interlaver regions successfully. The progressive contraction of the basal spacings of the structure upon increased oxidation implies the occurrence of interlayer polymerization.

The UV-vis spectra give strong evidence for the intercalation and polymerization of metanilic anion in NiAl LDH and provide information regarding the oxidation states of the interlayer products. The absorption bands centered at 300 and 550 nm are indicative of  $\pi_{\rm B} - \pi_{\rm B}^*$  and  $\pi_{\rm Q} - \pi_{\rm B}^*$  transitions, respectively. The hypsochromic shift of the  $\pi_{\rm Q}$ - $\pi_{\rm B}$ <sup>\*</sup> transition absorption indicates that the oxidation state of the interlayer my position and the oxidation state of the interlayer<br>indicates that the oxidation state of the interlayer<br>product was increased remarkably with an increase in<br>the quantity of oxidizing agent. The characteristic<br>absorption indicates that the community state of the internet the quantity of oxidizing agent. The characteristic absorption bands of the aniline and sulfonic anions in FTIR spectra demonstrate the intercalation of metanilic anion compared with the  $NiAl-NO<sub>3</sub>$  LDH precursor. The changes in the characteristic absorption bands assigned to the vibration of  $-B-$  and  $-Q-$  indicate that the oxidative degree of the polymerization products of the polymerization products<br>increasing quantity of oxidizing<br>tron spectroscopy was used to<br>iical compositions, the chemical our and the products produced products increased along with increasing quantity of ovidizing agent  $(NH_4)$ ,  $S_2O_8$ .

X-ray photoelectron spectroscopy was used to determine the chemical compositions, the chemical determine the chemical components, the chemical environment changes of the N atom and the  $\frac{1}{4}$ 



Figure 7. The interlayer oxidative polymerization of the metanilic monomer by the oxidizing agent  $(NH_4)_2S_2O_8$ .

evaluations of the various oxidation states of interlayer<br>PANIS. The oxidation level  $[=N-]/[-NH-]$  and the protonic ratio  $[N^+]/[N]$  with increased treatment with  $(NH_4)_2S_2O_8$  confirms that the quantity of oxidizing agent has a significant effect on the oxidation states of the polymerization products. In addition, the decreas  $(NH_4)_2S_2O_8$  confirms that the quantity of oxidizing agent has a significant effect on the oxidation states of the polymerization products. In addition, the decrease of the relative atomic concentration of N along with the increase of S indicates that  $S_2O_8^{2-}$  anions exchanged for  $NO<sub>3</sub><sup>-</sup>$  anions in the LDH interlayer and initiated the <sup>8</sup> anions exchanged for<br>
rlayer and initiated the<br>
nonomer. Therefore, the<br>
f metanilic anion in a NO<sub>3</sub> anions in the LDH interlayer and initiated the<br>polymerization of interlayer monomer. Therefore, the<br>study of the polymerization of metanilic anion in a<br>restricted environment between LDH sheets may have study of the polymerization of metanilic anion in a restricted environment between LDH sheets may have applications for a novel nanoreaction vessel for confined chemical reactions.

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