

# IDENTIFICATION OF NEOFORMED Ni-PHYLLOSILICATES UPON Ni UPTAKE IN MONTMORILLONITE: A TRANSMISSION ELECTRON MICROSCOPY AND EXTENDED X-RAY ABSORPTION FINE STRUCTURE STUDY

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**Abstract**—The aim of this work was to investigate whether neoformed Ni-phyllsilicates can be observed and identified using transmission electron microscopy (TEM) in combination with energy dispersive spectroscopy (EDS). The investigations focused on Ni-phyllsilicates formed from Ni-doped montmorillonite. The reaction conditions (pH 8,  $[\text{Ni}]_{\text{initial}} = 660$  and  $3300 \mu\text{M}$ ,  $0.2 \text{ M Ca}(\text{NO}_3)_2$ ) employed were similar to those used in previous polarized extended X-ray absorption fine structure (P-EXAFS) investigations of neoformed Ni-phyllsilicates in a Ni-montmorillonite system.

The TEM investigations of Ni-doped montmorillonite revealed the presence of small, thin particles consisting of coherent stacks that yielded only three to five lattice fringes with spacings consistent with smectites. These small particles were neoformed phyllsilicates, based on the fact that the small particles were only observed in Ni-doped samples and their Ni content, as determined from EDS analysis, was high (up to 10 wt.% NiO). Furthermore, the particles did not possess the characteristic properties of montmorillonite particles, such as a 2:1 Si to Al ratio; instead these particles were rich in Si (up to 75 wt.%  $\text{SiO}_2$ ). Unlike montmorillonite, these particles did not contain any Fe. The particles were also significantly more resistant to electron beam damage than montmorillonite particles, and EXAFS measurements confirmed the presence of neoformed Ni-phyllsilicates.

The TEM study further indicates the presence of a variety of additional minerals (*e.g.* cristobalite, halloysite) and an X-ray amorphous Si-rich phase. A Ni signal could only be detected in the latter phase at high Ni loadings ( $403 \mu\text{mol/g}$ ), suggesting that Ni uptake at low metal loadings ( $<90 \mu\text{mol/g}$ ) is mainly controlled by the neoformation of phyllsilicates.

**Key Words**—EXAFS, Montmorillonite, Neoformed Phyllsilicates, Ni, TEM.

## INTRODUCTION

Using extended X-ray absorption fine structure (EXAFS) spectroscopy, it has been demonstrated that Ni-, Co- and Zn-containing precipitates can form when clay minerals and Al and Si (hydr)oxides are reacted with aqueous solutions containing Ni, Co and Zn, at initial metal concentrations below saturation of the pure metal (hydr)oxide forms (Towle *et al.*, 1997; Scheidegger *et al.*, 1998; Manceau *et al.*, 1999; Thompson *et al.*, 1999a; Thompson *et al.*, 1999b; Morton *et al.*, 2001; Schlegel *et al.*, 2001; Dähn *et al.*, 2002; Lee *et al.*, 2004). For example, Scheidegger *et al.* (1998) observed the formation of a Ni-Al layered double hydroxide (Ni-Al LDH) phase when pyrophyllite (a 2:1 clay mineral) was doped with Ni. The Ni-Al LDH phase formed after a contact time of only a few minutes, indicating that nucleation of the new phase occurs rapidly. Using polarized-EXAFS (P-EXAFS) Dähn *et al.*

(2002) and Schlegel *et al.* (2001) showed the presence of neoformed phyllsilicates after treating di- and tri-octahedral 2:1 clay minerals (montmorillonite, hectorite) with Ni and Zn solutions at elevated pH and metal concentrations. The neoformed phases were oriented parallel to the octahedral smectite sheets (Schlegel *et al.*, 2001; Dähn *et al.*, 2002). In the case of Schlegel *et al.* (2001) a structural link of the neoformed phases to hectorite particles could be observed.

Investigations by TEM supported the evidence from EXAFS studies that precipitates formed upon metal uptake by clay minerals. For example, TEM was used to confirm the formation of a Ni-Al LDH phase upon Ni uptake in the presence of pyrophyllite (Scheidegger *et al.*, 1996b). The authors observed surface deposits on pyrophyllite particles doped with Ni at  $\text{pH} > 7$ ; these deposits were not present in the undoped sample and occurred preferentially along the edges of the pyrophyllite particles. Similarly, Thompson *et al.* (1999b) used TEM to discern the presence of a Co-Al LDH phase formed upon Co uptake in the presence of kaolinite (a 1:1 clay mineral) at  $\text{pH} 7.8$ . Nanometer-sized crystalline particles associated with the edges and basal surfaces of kaolinite were observed in the Co-doped material.

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In this study, TEM imaging and energy dispersive spectroscopy (EDS) analysis were conducted to investigate whether Ni-phyllsilicates form in the presence of Ni-doped montmorillonite. The reaction conditions used in this study (pH 8,  $[\text{Ni}]_{\text{initial}} = 660$  and  $3300 \mu\text{M}$ ,  $0.2 \text{ M Ca}(\text{NO}_3)_2$ ) were similar to those used in a previous P-EXAFS study which indicated neof ormation of a Ni-phyllsilicate phase (Dähn *et al.*, 2002). In contrast to the previous experiments by Dähn *et al.* (2002), this TEM/EDS study was conducted with 'as received' montmorillonite without any further purification and conditioning procedures (untreated). Untreated montmorillonite was used since smectite particles in this material were more resistant to electron beam damage compared to those in purified and conditioned (Ca- and Na-treated) montmorillonite. The crystallinity of smectite particles of the treated samples suffered beam damage within  $<1 \text{ s}$ , whereas the untreated material could withstand the electron beam for several seconds and up to 1 min. Complementary EXAFS measurements were performed with both untreated and treated montmorillonite to check whether the Ni-uptake mechanism (neof ormation of a Ni-phyllsilicate) remained unchanged by using the untreated clay.

## MATERIALS AND METHODS

### Montmorillonite characterization

The STx-1 montmorillonite used was purchased from the Source Clays Repository of The Clay Minerals Society. X-ray diffraction (XRD) investigations of the 'as received' montmorillonite reveal the presence of minor quantities of calcite, quartz and kaolinite amounting to  $<1 \text{ wt.}\%$ . For the TEM investigations, the untreated clay material was used, whereas EXAFS measurements were performed on both untreated and treated clay material. In the treatment process, the montmorillonite was washed thoroughly three times with  $1 \text{ M NaClO}_4$  to produce Na-montmorillonite. The  $<0.5 \mu\text{m}$  size fraction was separated by repeated washing with de-ionized water, combined with centrifugation ( $600 \text{ g max.}$ ,  $\sim 7 \text{ min}$ ) and decantation of the supernatant suspensions immediately followed by flocculation with  $1 \text{ M NaClO}_4$  (Baeyens and Bradbury, 1997). Soluble hydroxy-Al compounds were removed by an acid treatment (pH 3.5, for 1 h) followed by neutralization (pH 7).

Conversion to Ca-montmorillonite was achieved by dialysis methods. Visking dialysis membranes (Medical Int., London) were partially filled with Na-montmorillonite suspensions, sealed and then agitated in a 1 L container filled with a  $0.2 \text{ M Ca}(\text{NO}_3)_2$  solution. The background electrolyte was refreshed daily, and the procedure was continued until the electrical conductivity of the solution was equal to that of the original  $0.2 \text{ M Ca}(\text{NO}_3)_2$ . The final conditioned montmorillonite suspension was stored at  $4^\circ\text{C}$  in the dark to minimize microbial growth.

The cation exchange capacity (CEC) of the untreated STx-1 montmorillonite is  $844 \text{ meq kg}^{-1}$  (Van Olphen and Fripiat, 1979). The CEC of the treated Ca-montmorillonite was determined to be  $970 \text{ meq kg}^{-1}$  by using the isotopic dilution method with  $^{45}\text{Ca}$  (Baeyens and Bradbury, 1995). The external surface area of the untreated and treated STx-1 montmorillonite was measured by the  $\text{N}_2$ -BET technique to be  $82 \text{ m}^2\text{g}^{-1}$  and  $89 \text{ m}^2\text{g}^{-1}$ , respectively. These values are similar to that of  $84 \text{ m}^2\text{g}^{-1}$  given by Van Olphen and Fripiat (1979).

### Sample preparation

The montmorillonite samples with low and high Ni loadings were prepared for analysis by TEM and EXAFS. Samples with low Ni loading were prepared by adding  $0.64 \text{ g}$  of the untreated STx-1 to  $40 \text{ mL}$  of a  $0.2 \text{ M Ca}(\text{NO}_3)_2$  solution. The pH was adjusted to 8 and  $60 \text{ mL}$  of a Ni solution of  $1100 \mu\text{M}$  (pH 8.0,  $0.2 \text{ M Ca}(\text{NO}_3)_2$ ) was added, resulting in a solid to liquid ratio of  $6.4 \text{ g/L}$  and an initial Ni concentration ( $[\text{Ni}]_{\text{initial}}$ ) of  $660 \mu\text{M}$ . A previous study (Dähn *et al.*, 2002) demonstrated that the Ni solution was stable over a time period of up to 1 year and thus Ni uptake was not due to  $\text{Ni}(\text{OH})_2(\text{s})$  precipitation in this study. The high ionic strength  $\text{Ca}(\text{NO}_3)_2$  background electrolyte was used to block Ni uptake on cation exchange sites. The experiments were conducted in a glove box under  $\text{N}_2$  atmosphere ( $\text{CO}_2$  and  $\text{O}_2 < 2 \text{ ppm}$ ). The samples were centrifuged after reaction times of 14 (sample A; Table 1) and 60 days (sample B; Table 1). The supernatant solution was separated for analysis and the wet pastes were placed into Plexiglas holders, sealed and stored in a refrigerator to keep them moist prior to TEM and EXAFS measurements. The inductively coupled plasma-optical emission spectroscopy (ICP-OES) measurements of the supernatants showed that the quantity of Ni taken up varied from  $43 \mu\text{mol/g}$  (sample A; Table 1) to  $90 \mu\text{mol/g}$  (sample B; Table 1).

A sample with high Ni loading (sample C; Table 1) was prepared by adding  $0.64 \text{ g}$  of untreated STx-1 to  $95 \text{ mL}$  of  $0.2 \text{ M Ca}(\text{NO}_3)_2$ . The pH was adjusted to 8, and  $5 \text{ mL}$  of a Ni solution ( $[\text{Ni}] = 6.6 \times 10^{-2} \text{ M}$ , pH 3.0,  $0.2 \text{ M Ca}(\text{NO}_3)_2$ ) were added. After 60 days of reaction the proportion of Ni taken up amounted to  $403 \mu\text{mol/g}$  (sample C, Table 1).

An additional sample with a low Ni loading (sample D; Table 1) was prepared for EXAFS measurements using treated Ca-montmorillonite ( $<0.5 \mu\text{m}$  size fraction). Except for the solid to liquid ratio ( $5.2 \text{ g/L}$ ) the reaction conditions were identical to those described above for sample A (Table 1).

In addition to Ni, the supernatant solutions of the Ni-doped montmorillonite samples were also analyzed for Si and Al by ICP-OES (Table 1). The Al concentration in the samples was too low ( $<6 \mu\text{M}$ ) to produce reliable ICP-OES measurements. The Si concentration in the

Table 1. Samples analyzed by TEM and EXAFS for untreated and treated Ni-doped montmorillonite.

Samples	Clay material	Clay treatment	Method	Solid/liquid ratio (g/L)	Ni uptake (%)	[Ni] ( $\mu\text{mol/g}$ )	Reaction time (days)	[Si] ( $\mu\text{M}$ )
A	montmorillonite	untreated	TEM, EXAFS	6.4	42 <sup>a</sup>	43	14	406
B	montmorillonite	untreated	TEM	6.4	89 <sup>a</sup>	90	60	527
C	montmorillonite	untreated	TEM	6.4	80 <sup>b</sup>	403	60	488
D	montmorillonite	treated	EXAFS	5.2	32 <sup>a</sup>	40	14	282

<sup>a</sup> Initial Ni concentration = 660  $\mu\text{M}$

<sup>b</sup> Initial Ni concentration = 3300  $\mu\text{M}$

Ni-doped montmorillonite samples varied between 282 and 527  $\mu\text{M}$ .

### TEM/EDS

The TEM investigations were performed using a JEOL 2000 FX transmission electron microscope operated at 200 kV at CEA-Cadarache in France. Energy dispersive spectroscopy (EDS) was performed using an Oxford Instruments Si-Li detector in convergent beam mode with a beam size of  $\sim 10$  nm. The count time for the EDS analysis was 60 s with a deadtime of 0–10% and a count rate of 500 to 2000 counts/s. The detection limit of the EDS was 0.1 wt.% (Buseck and Self, 1992). The EDS data were semi-quantitatively analyzed using the program TEMQUANT from ISIS (Table 2).

The TEM/EDS samples were prepared by diluting a small amount of the centrifuged wet montmorillonite pastes in de-ionized water and by transferring a drop (10  $\mu\text{L}$ ) of the diluted suspension to a gold mesh grid with a holey carbon support film. The drop was then allowed to dry over 24 h in air at a temperature of 22°C and a relative humidity of 70% prior to the TEM investigations.

### EXAFS data collection and reduction

The Ni K-edge EXAFS spectra were obtained at the Swiss-Norwegian beamline (SNBL) at the European Synchrotron Radiation Facility (ESRF, Grenoble, France). All spectra were recorded in fluorescence mode at room temperature using a Si(111) monochromator and a Stern-Heald-type detector (Lytle detector,

The EXAFS Co.). Higher-order harmonics were suppressed by using an Au-coated mirror. The monochromator angle was calibrated by assigning 8333 eV to the first inflection point of the K-absorption edge spectrum of Ni metal. Several EXAFS scans were averaged to improve the signal-to-noise ratio.

Data reduction was carried out with the WinXAS 3.1 software package (Ressler, 1998). The energy was converted to photoelectron wave vector units ( $\text{\AA}^{-1}$ ) by assigning the origin  $E_0$  to the first inflection point of the absorption edge. Radial structure functions (RSFs) were obtained by Fourier transforming  $k^3$ -weighted  $\chi(k)$  functions between 3.2 and 10  $\text{\AA}^{-1}$  using a Bessel window function with a smoothing parameter of 4. Amplitude and phase-shift functions were calculated with FEFF 8.0 (Rehr *et al.*, 1991) using the structures of  $\beta$ -Ni(OH)<sub>2</sub> and Ni-Talc (Perdikatsis and Burzlaff, 1981) as references. Fits were performed in  $R$  space in the 0.6 to 3.5  $\text{\AA}$  interval. The amplitude reduction factor ( $S_0^2$ ) was determined to be 0.85 from the experimental  $\beta$ -Ni(OH)<sub>2</sub> EXAFS spectrum. The deviation between the fitted and the experimental spectra (%Res) is given by:

$$\% \text{Res} = \frac{\sum_{i=1}^N |y_{\text{exp}}(i) - y_{\text{theo}}(i)|}{\sum_{i=1}^N y_{\text{exp}}(i)}$$

where  $N$  is the number of points in the fit window, and  $y_{\text{exp}}$  and  $y_{\text{theo}}$  are the experimental and theoretical RSF values. The precision for the bond distances ( $R$ ) was

Table 2. EDS analyses of the montmorillonite samples A–C normalized to 100 wt.%.

	SiO <sub>2</sub>	Al <sub>2</sub> O <sub>3</sub>	Fe <sub>2</sub> O <sub>3</sub>	NiO	MgO	CaO	Total
EDS1	94.28	4.45	0.24		0.59	0.44	100
EDS2	58.63	40.96			0.10	0.31	100
EDS3	92.94	4.94			1.2	0.92	100
EDS4	70.06	23.03	0.54		5.27	1.1	100
EDS5	70.65	21.60	0.27	1.09	4.07	2.32	100
EDS6	75.14	10.06		10.40	2.30	2.10	100
EDS7	64.10	19.51	0.48	9.08	4.54	2.29	100
EDS8	70.32	18.87		5.22	3.96	1.63	100
EDS9	58.65	39.85			0.66	0.84	100

previously estimated to be  $\pm 0.02$  Å for  $R_{\text{Ni-O}}$ , and  $\pm 0.03$  Å for  $R_{\text{Ni-O}}$  and  $R_{\text{Ni-Si}}$ , and  $\pm 0.5$  for the coordination numbers (Dähn *et al.*, 2002).

## RESULTS

### TEM observations

**Starting material.** Before studying Ni-doped montmorillonite samples, we investigated the untreated clay material. The TEM observations demonstrated that the untreated starting material consisted mainly of smectite particles. However, additional mineral phases were identified. The variety of mineral phases present in the untreated montmorillonite material is demonstrated in Figure 1 in a low-magnification image. The image shows cauliflower-like features which are characteristic of cristobalite (cb) (Jackson *et al.*, 1977). Analysis by EDS showed that the cauliflower-like features consist of  $\sim 94$  wt.%  $\text{SiO}_2$  and  $\sim 4$  wt.%  $\text{Al}_2\text{O}_3$  (Table 2, EDS1), consistent with the composition of cristobalite ( $\text{SiO}_2$ ; Strunz and Nickel, 2001). Based on the mineral morphology, the long ( $\sim 900$  nm) columnar particle at the right hand side of the image was identified as halloysite (ha) (Mitra and Bhattacharjee, 1975; Kirkman,

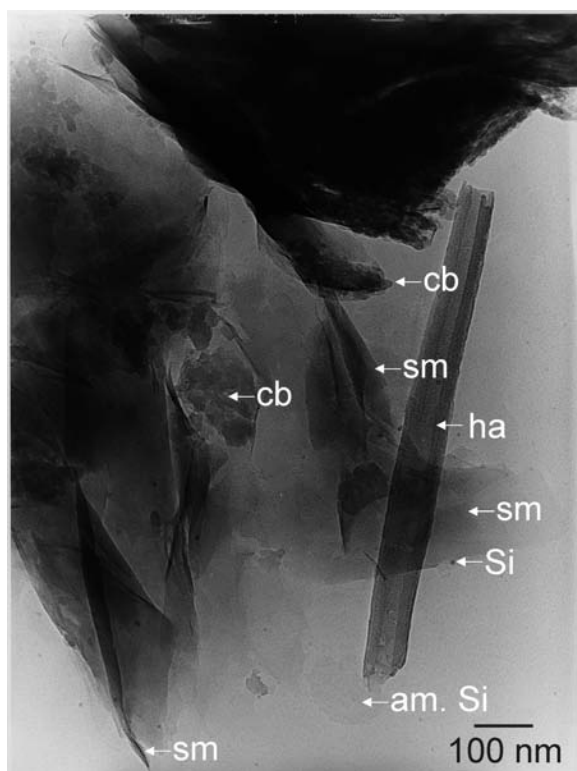


Figure 1. TEM image of different mineral phases in untreated montmorillonite. Only a few of the smectite particles (sm) observed in the image are labeled. The cauliflower-like features are cristobalite (cb), dark spots are clusters of Si (Si), the long particle is halloysite (ha) with an X-ray amorphous Si-rich phase (am. Si) at the lower tip.

1981). Halloysite is a 1:1 clay mineral and is the hydrated form of kaolinite. The halloysite crystal had a fiber-like structure (diameter  $\sim 70$  nm) and was hollow. The EDS analysis showed that the halloysite crystal consists of  $\sim 59$  wt.%  $\text{SiO}_2$  and  $\sim 41$  wt.%  $\text{Al}_2\text{O}_3$  (Table 2, EDS2). This finding is consistent with the structural formula of halloysite ( $\text{Al}_4(\text{OH})_8(\text{Si}_4\text{O}_{10}) \cdot 2\text{H}_2\text{O}$ ; Strunz and Nickel, 2001). On the lower tip of the halloysite particle there was a diffuse and cloudy feature identified as an X-ray amorphous Si-rich phase consisting of  $\sim 94$  wt.%  $\text{SiO}_2$  and  $\sim 5$  wt.%  $\text{Al}_2\text{O}_3$  (Table 2, EDS3).

An image of a smectite particle is presented in Figure 2a. At the lower edge of this particle, lattice fringes are visible (boxed areas in Figure 2). Lattice fringes are observed when the edges of smectite particles are curved upwards. If this is the case, the layers of smectite particles are parallel to the beam, allowing the observation of (001) lattice fringes. The lattice spacing of the illustrated particle (12 Å, Figure 2b,c) is characteristic of the spacing between (001) silicate layers in smectites ( $d_{001} = 12$  Å, Meike, 1989; Allen, 1992). Only small areas of the smectite particle were in appropriate orientations to show clear lattice-fringe images (Figure 2b,c) indicating highly disordered stacking of the crystallites. The selected area electron diffraction (SAED) patterns of such intergrowths contained diffuse reflections resulting from a mixture of several disordered crystallites (Figure 2d; Peacor, 1992; Dong and Peacor, 1996). The EDS analysis showed that the core of the particle consists of  $\sim 70$  wt.%  $\text{SiO}_2$ ,  $\sim 23$  wt.%  $\text{Al}_2\text{O}_3$ ,  $\sim 0.5$  wt.% FeO,  $\sim 5$  wt.% MgO and  $\sim 1$  wt.% CaO (Table 2, EDS4) which is consistent with montmorillonite. The montmorillonite particles in the untreated material were generally very fragile and suffered from radiation damage after exposing the sample to the electron beam for  $< 1$  min (*i.e.* the lattice fringes increasingly smeared out and finally disappeared). The image also shows the presence of an X-ray amorphous Si-rich phase (see above) associated with the smectite particle (dotted circled area in Figure 2).

**Ni-doped montmorillonite.** In the following we discuss whether TEM images and EDS analyses reveal any changes when the untreated clay material was doped with Ni. Sample B with low Ni loading consists of an assembly of smectite, cristobalite and halloysite particles and an X-ray amorphous Si-rich phase (Figure 3, Table 1). The TEM images show the presence of the same mineral phases as observed in the untreated material (Figure 1).

The EDS analysis of a series of cristobalite and halloysite particles and X-ray amorphous Si-rich phases did not detect any Ni signals in samples with low Ni loadings (samples A and B, Table 1). We could, however, obtain a Ni signal when smectite particles were analyzed. The textures from which these analyses

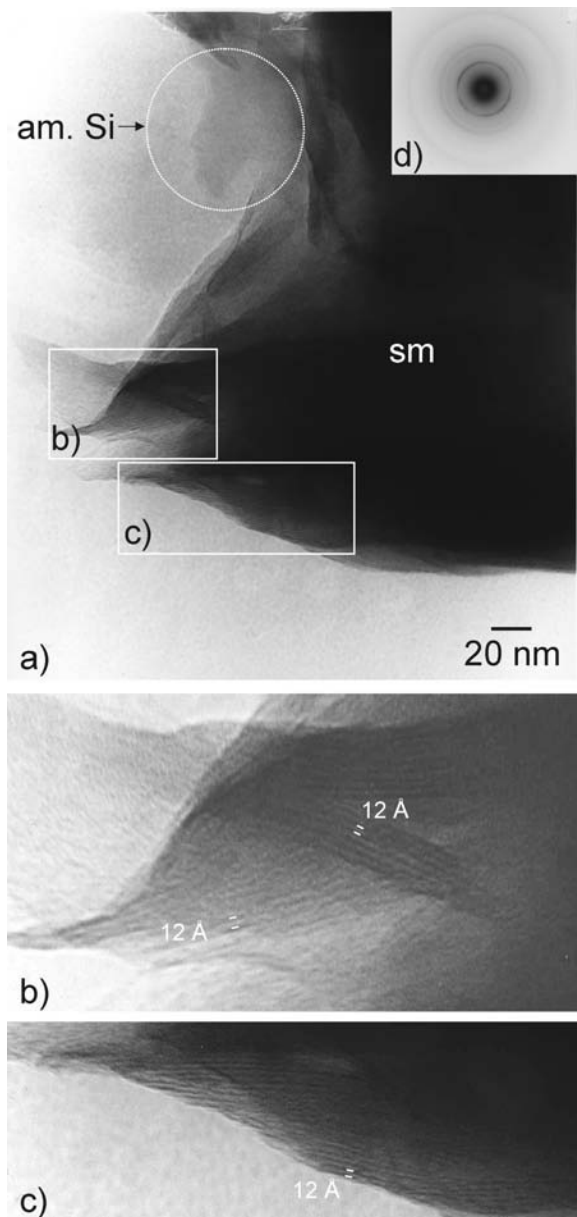


Figure 2. (a) Higher-resolution TEM image of a smectite (sm) in untreated montmorillonite with an X-ray amorphous Si-rich phase (am. Si, dotted circular area) and lattice fringes (boxed areas). (b,c) Magnified areas of Figure 2a. (d) SAED pattern of the smectite particle.

were obtained are shown in Figure 4. While the X-ray amorphous Si-rich phase and cristobalite contained no Ni, EDS analysis indicates that the center of the smectite (circled area) contained  $\sim 1$  wt.% NiO (Table 2, EDS5).

Several montmorillonite particles contained a significantly larger Ni content (up to 10 wt.% NiO). The observed particles were generally very small and yielded only a few lattice fringes. Figure 5 shows a high-magnification image of particles (sm1, sm2, sm3) with 3–5 coherent lattice fringes (sample B, Table 1) and the

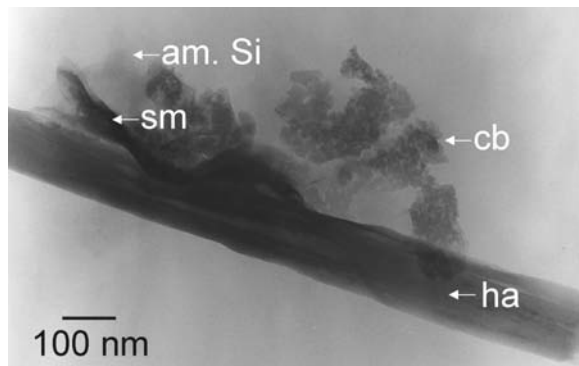


Figure 3. Halloysite crystal (ha), cristobalite (cb), X-ray amorphous Si-rich phases (am. Si) and smectite (sm) in a Ni-doped montmorillonite sample ( $90 \mu\text{mol/g}$ , 60 days' reaction time, sample B).

layer spacing of the lattice fringes in these particles is 11–12 Å, consistent with montmorillonites. Smectite particles with only a few lattice fringes were not observed in the starting material, suggesting that their appearance is related to the presence of Ni or a disruption of particles during exchange.

The EDS analysis of the very fine tip of sm3 (circled area in Figure 5) revealed that the tip of this smectite particle contained  $\sim 75$  wt.%  $\text{SiO}_2$ ,  $\sim 10$  wt.%  $\text{Al}_2\text{O}_3$  and  $\sim 10$  wt.% NiO (Table 2, EDS6). The fact that the small particle (sm3) contained large amounts of NiO and  $\text{SiO}_2$

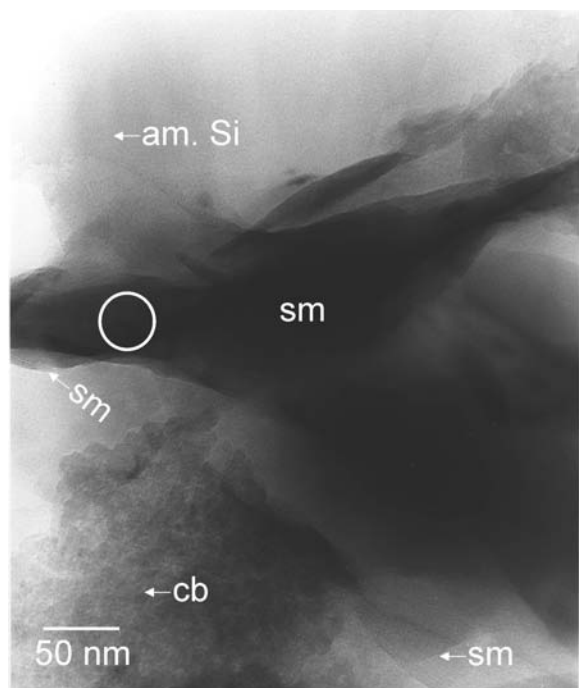


Figure 4. Smectite (sm), cristobalite (cb) and an X-ray amorphous Si-rich phase in a Ni-doped montmorillonite sample ( $43 \mu\text{mol/g}$ , 14 days' reaction time, sample A). The circled area marks the area of the EDS analysis of the smectite.

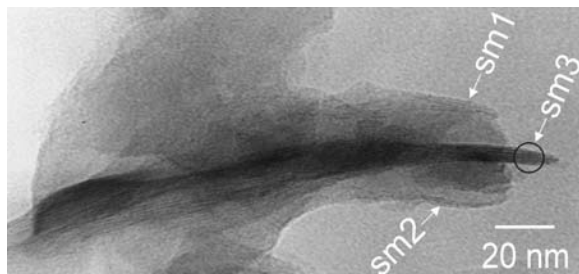


Figure 5. Neoformed phyllosilicate (sm1–sm3) in a Ni-doped montmorillonite sample (90  $\mu\text{mol/g}$ , 60 days' reaction time, sample B). The circled area marks the area of the EDS analysis of sm3.

and relatively small amounts of  $\text{Al}_2\text{O}_3$  demonstrates the presence of a Ni-rich silicate phase. The EDS analysis of the tip of sm3 further indicates that this smectite particle lacked Fe. For comparison, in the starting clay material, the Fe content in smectites, as measured by EDS, varied between 0.1 and 1 wt.%. A further characteristic of the Ni-rich small particles was their resistance to electron beam damage. While smectite particles in the starting material were generally very fragile and suffered from radiation damage after exposure to the electron beam for times of <1 min, the Ni-rich particles could withstand the electron beam for a much longer period (> 5 min).

The TEM/EDS investigations with samples having a high Ni loading (sample C, Table 1) again revealed no Ni signal from cristobalite and halloysite particles. In these samples, however, Ni was detected not only in the

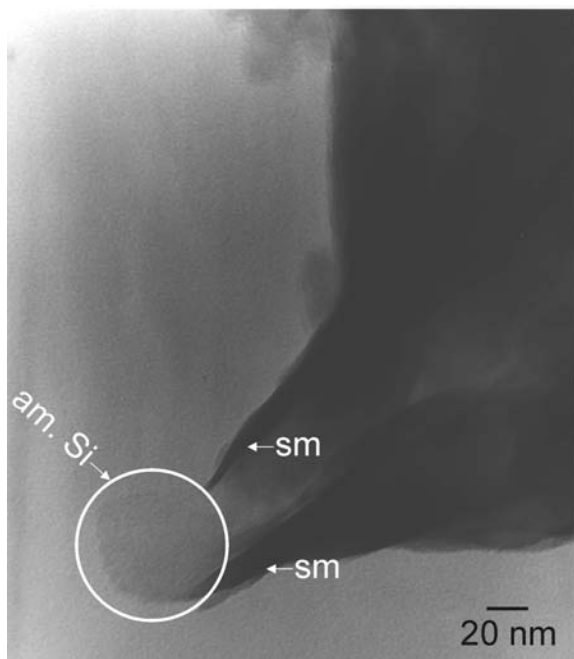


Figure 6. Smectite (sm) and an X-ray amorphous Si-rich phase (circled area) in a Ni-doped montmorillonite sample (403  $\mu\text{mol/g}$ , 60 days' reaction time, sample C).

smectite, but also in the X-ray amorphous Si-rich phases. Figure 6 shows a high-magnification image of an X-ray amorphous Si-rich phase (circled area) sandwiched between two smectite regions. Whereas the tip of the lower smectite particle contained ~9 wt.% NiO (Table 2, EDS7), EDS data showed that the X-ray amorphous Si-rich phase contained ~5 wt.% NiO (Table 2, EDS8).

Figure 7 shows an image of a halloysite particle present in the Ni-doped material (sample A, 43  $\mu\text{mol/g}$ , reaction time: 14 days). The EDS data analysis showed that the halloysite particle contained no detectable Ni (Table 2, EDS9). The diameter of the fiber is ~150 nm and exhibited holes with diameters of between 8 and 20 nm. No such holes occurred in halloysite particles in the starting material (Figure 1). Whereas halloysite particles were still observable when the material was doped with Ni at near neutral conditions for 14–60 days, no halloysite particles could be observed when the material underwent the purification and conditioning process (see the section on sample preparation).

#### EXAFS

A previous P-EXAFS study has shown unambiguously that Ni uptake in the presence of montmorillonite under similar reaction conditions resulted in the neof ormation of a Ni-phyllosilicate (Dähn *et al.*,



Figure 7. Halloysite crystal with core holes in a Ni-doped montmorillonite sample (43  $\mu\text{mol/g}$ , 14 days' reaction time, sample A).

Table 3. Structural information derived from the EXAFS analysis using a three-shell fit approach for 14 days of reaction time ([Ni] = 43  $\mu\text{mol/g}$ , sample A, untreated, and [Ni] = 40  $\mu\text{mol/g}$ , sample D, treated montmorillonite).

Sample	Ni–O			Ni–Ni			Ni–Si			$\Delta E_0$ (eV)	%Res
	CN <sub>Ni–O</sub>	$R_{\text{Ni–O}}$ (Å)	$\sigma^2$ (Å <sup>2</sup> )	CN <sub>Ni–Ni</sub>	$R_{\text{Ni–Ni}}$ (Å)	$\sigma^2$ (Å <sup>2</sup> )	CN <sub>Ni–Si</sub>	$R_{\text{Ni–Si}}$ (Å)	$\sigma^2$ (Å <sup>2</sup> )		
Untreated	5.6	2.05	0.006	4.0	3.09	0.007	4.1	3.26	0.007 <sup>f</sup>	0.6	3.6
Treated	5.7	2.05	0.006	3.1	3.09	0.006	3.5	3.28	0.006 <sup>f</sup>	0.8	3.7

CN, R,  $\sigma^2$ ,  $\Delta E_0$  are the coordination numbers, interatomic distances, Debye-Waller factors and inner potential corrections. <sup>f</sup>: fixed to the value obtained for Ni–Ni

2002). While in the previous Ni uptake study, treated montmorillonite (STx-1) was used, the TEM investigations here were performed with the untreated form of the same material.

We performed EXAFS experiments with Ni-doped untreated and treated montmorillonite to check whether the use of treated material modified the Ni uptake mechanism. The untreated material also contains particles >0.5  $\mu\text{m}$  and minor minerals such as calcite, cristobalite, halloysite and X-ray amorphous Si-rich phases. The  $k^3$ -weighted EXAFS spectra for both montmorillonite samples doped with Ni at pH 8 for a reaction time of 14 days are shown in Figure 8 as an example (samples A and D, Table 1). The figure shows that the differences in the spectra attributable to the treatment procedure are minor. The multifrequency wave shape suggests the presence of at least two backscattering pairs. The corresponding experimental and the simulated Fourier transforms (FTs) are illustrated in Figure 9. The amplitude and position of the first RSF peaks (Ni–O contribution) do not differ in the two samples. The position of the second RSF peak for both samples is identical ( $R + \Delta R \approx 2.75$  Å) and characteristic for Ni–Ni pairs (Manceau and Calas, 1986; Manceau, 1990; Pandya *et al.*, 1990; Scheidegger *et al.*, 1996a, 1997, 1998; Scheinost *et al.*, 1999;

Mansour and Melendres, 1998; Dähn *et al.*, 2002). The amplitude of the second RSF peak, however, is slightly higher in the untreated sample compared to the treated sample.

Structural parameters obtained by data analysis are listed in Table 3. Data analysis indicates that the first shell consists of  $5.6 \pm 0.5$  Ni–O pairs at a bond distance (2.05 Å) which is typical of six-fold coordinated Ni (Pandya *et al.*, 1990). In both samples, Ni–Ni and Ni–Si distances were observed ( $R_{\text{Ni–Ni}} = 3.09$  Å and  $R_{\text{Ni–Si}} = 3.26$ – $3.28$  Å) which match well with those in Ni-phyllsilicates ( $R_{\text{Ni–Ni}} = 3.05$ – $3.09$  Å,  $R_{\text{Ni–Si}} = 3.26$ – $3.27$  Å; Manceau and Calas, 1986; Charlet and Manceau, 1994; Dähn *et al.*, 2002). This finding confirms the formation of a Ni-phyllsilicate phase in both samples and that the use of the untreated clay did not modify the predominant uptake mechanism under the reaction conditions used.

## DISCUSSION

### Identification of Ni-phyllsilicates

The TEM images of both the starting material and Ni-doped montmorillonite demonstrated the presence of smectite particles with a shape and lattice fringes

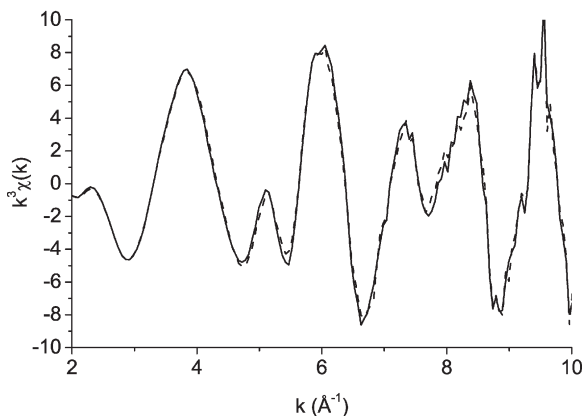


Figure 8.  $k^3$ -weighted Ni K edge EXAFS spectra of montmorillonite doped with Ni for 14 days for the untreated (solid line, [Ni] = 43  $\mu\text{mol/g}$ , sample A) and the treated (dashed line, [Ni] = 40  $\mu\text{mol/g}$ , sample D) clay material at pH 8.

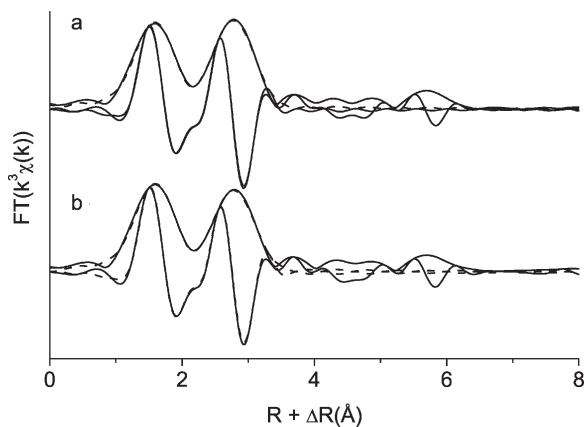


Figure 9. Experimental and theoretical Fourier transforms (modulus and imaginary parts, solid line: experimental data, dotted line: least-squares fit) obtained from the EXAFS spectra presented in Figure 8; (a) untreated and (b) treated.

characteristic of smectite ( $d_{001} = 12 \text{ \AA}$ ; Meike, 1989; Allen, 1992). The  $d$  spacing varies with the hydration level and is  $10 \text{ \AA}$  for a fully dehydrated smectite, and  $15\text{--}20 \text{ \AA}$  for hydration levels between 10 and 50% (Kraehenbuel *et al.*, 1987). A lattice spacing of  $12 \text{ \AA}$  refers to hydration levels of smectite particles between 5 and 10%. The study of Kraehenbuel *et al.* (1987) further demonstrated that a lattice spacing of  $12 \text{ \AA}$  indicates the presence of one layer of water between smectite layers, whereas a lattice spacing of  $15 \text{ \AA}$  is characteristic of two layers of water. Thus, although the TEM samples were measured under vacuum conditions in the TEM, a monolayer of water still remained in the interlayer of the smectites. The finding that lattice fringes were observable in the TEM images suggests that the crystallinity of the smectite particles did not significantly change during the Ni-uptake experiments.

In the Ni-doped clay material we also observed small, thin particles with only 3–5 lattice fringes (lattice spacing  $\sim 11\text{--}12 \text{ \AA}$ ). There are several arguments indicating that these small particles (3–5 lattice fringes) are neoformed phyllosilicates. First of all, similar small particles were not observed in the starting clay material, and although these particles are very small in size, their Ni content ( $\sim 10 \text{ wt.}\% \text{ NiO}$ ; sample B, Figure 5) is significantly higher than the average Ni content in the sample B of  $0.7 \text{ wt.}\% \text{ NiO}$ . The Ni content of these small particles is also significantly higher than that of the individual Ni-doped montmorillonite particles, which never exceeded  $4 \text{ wt.}\% \text{ NiO}$  (sample B). Furthermore, EDS analysis (Table 2, EDS6) confirmed that the Ni-doped montmorillonite particles are consistent with a structural formula for smectites (2:1 ratio of Si to Al), whereas the small particles are Si rich ( $\sim 75 \text{ wt.}\% \text{ SiO}_2$ ) and Al poor ( $\sim 10 \text{ wt.}\% \text{ Al}_2\text{O}_3$ ). Moreover, EDS analysis showed that the Fe content in the Ni-doped montmorillonite particles varied between 0.1 and  $1 \text{ wt.}\% \text{ (FeO)}$ , while the small particles are lacking Fe. This finding is consistent with the total FeO content of the STx-1 montmorillonite ( $0.4 \text{ wt.}\%$ ) (Dähn *et al.*, 2002) and is further evidence that these small particles were neoformed. Additional proof that the small particles were neoformed can be deduced from the fact that the small particles are significantly more resistant to electron beam damage than Ni-doped montmorillonite particles. Finally, the neoformation of a Ni-phyllosilicate is confirmed by the accompanying EXAFS measurements of this study which revealed structural parameters characteristic of Ni-phyllosilicates (Manceau and Calas, 1986; Charlet and Manceau, 1994; Dähn *et al.*, 2002).

Schlegel *et al.* (2001) investigated the uptake of Zn on hectorite and showed that neoformed phyllosilicates grow at the edges of smectite particles in continuity with the octahedral sheets. Using TEM we were not able to observe whether the neoformed Ni-phyllosilicates have a structural link with the montmorillonite particles. We

observed small Ni-rich particles identified as neoformed phyllosilicates either close to or at larger distances from Ni-doped montmorillonite particles. This is similar to the TEM findings by Thompson *et al.* (1999b). Investigating Co uptake on kaolinite, these authors identified the presence of Co-Al LDH precipitates that were not preferentially oriented with respect to the kaolinite particles, and some of the precipitates had no contact with kaolinite particles.

#### *Ni association with other mineral phases*

In all Ni-doped montmorillonite samples (samples A–C, Table 1), a Ni signal could be detected when probing smectite particles, but none of the other minor mineral phases present in the sorbent material (*e.g.* halloysite and cristobalite) contained Ni. The detection of no Ni signal from halloysite, the hydrated form of kaolinite, indicates that the neoformed Ni-phyllosilicates are mainly associated with montmorillonite particles. The only exception to this finding were X-ray amorphous Si-rich phases, which were observed to be associated with smectite particles throughout this TEM study. While no Ni signal could be detected upon examination of the X-ray amorphous Si-rich phases in samples with a low Ni loading (samples A and B, Table 1), the presence of a Ni signal (up to  $5 \text{ wt.}\% \text{ NiO}$ ) was detected in X-ray amorphous Si-rich phases in the sample with a high Ni loading (sample C, Table 1). The TEM/EDS findings thus indicate that Ni is primarily taken up by montmorillonite and neoformed Ni-phyllosilicates, and only a minor amount by the X-ray amorphous Si-rich phases.

It has been shown previously using EXAFS that Ni uptake on pyrophyllite and kaolinite at elevated Ni concentrations and pH resulted in the formation of a Ni-Al LDH phase (Scheidegger *et al.*, 1996a, 1996b, 1997). This TEM study confirms previous EXAFS studies demonstrating that Ni uptake on montmorillonite, pyrophyllite and kaolinite under similar reaction conditions results in the formation of different Ni phases (*e.g.* Scheidegger *et al.*, 1996a, 1996b, 1997; Dähn *et al.*, 2002). The precipitation of different Ni phases can be explained by different dissolution rates (Si release) of montmorillonite, pyrophyllite and kaolinite. Scheidegger *et al.* (1997) obtained Si release rates of  $4.6 \times 10^{-13} \text{ mol m}^{-2} \text{ s}^{-1}$  for Ni-doped pyrophyllite (pH 7.5, time range 0–216 h) and  $5.2 \times 10^{-12} \text{ mol m}^{-2} \text{ s}^{-1}$  for Ni-doped kaolinite (pH 7.5, time range 0–51 h). These Si release rates of Ni-doped pyrophyllite and kaolinite are at least one order of magnitude higher than the Si release rate of Ni-doped montmorillonite ( $3.5 \times 10^{-14} \text{ mol m}^{-2} \text{ s}^{-1}$ , pH 8.0, time range 1–206 days; Dähn *et al.*, 2002). The higher dissolution rates of pyrophyllite and kaolinite cause the rapid formation of an X-ray amorphous Al hydroxide phase, which is a necessary precursor for the formation of Ni-Al LDH (Thompson *et al.*, 1999b; Yamaguchi, *et al.*, 2001). In the case of Ni-doped

montmorillonite, the dissolution rate is much smaller compared to pyrophyllite and kaolinite and therefore it seems that an X-ray amorphous Al hydroxide phase, which would favor the formation of a Ni-Al LDH, did not form. This hypothesis is supported by this TEM study, which observed only the precipitation of X-ray amorphous Si-rich phases, whereas Thompson *et al.* (1999b) observed in a TEM study of kaolinite the precipitation of X-ray amorphous Al-rich phases. This explanation is in line with the observed differences between Ni-doped montmorillonite and Ni-doped pyrophyllite/kaolinite. While the formation of Ni-AL LDH is rapid ( $\geq 15$  min) in Ni-doped pyrophyllite and kaolinite samples (Scheidegger *et al.*, 1996a, 1996b, 1997), the formation of a Ni-phyllsilicate in Ni-doped montmorillonite is much slower ( $\geq 1$  day; Dähn *et al.*, 2002). It thus appears that when X-ray amorphous Al-rich phases are present in the Ni-clay system, the rapid formation of Ni-LDH phases is preferential, whereas when not enough Al is available, the formation of Ni-phyllsilicates is favored.

#### *Influence of the clay-purification and conditioning procedure*

In metal-uptake experiments with montmorillonite, purifying and conditioning procedures are commonly applied in order to separate the clay fraction ( $< 0.5 \mu\text{m}$ ), to convert the clay into the homoionic form and to remove impurities and soluble hydroxy-Al compounds. We have conducted TEM observations with both untreated and treated montmorillonite to investigate whether the various treatment steps did modify the morphology of the clay material. We observed that the crystallinity of the smectite particles decreased significantly with each purifying and conditioning step and that the smectite particles became more prone to radiation damage, even when the sample was exposed to the electron beam for only seconds. For these reasons the TEM investigations were conducted with untreated clay material.

To investigate the influence of purification and conditioning on the uptake process, accompanying EXAFS measurements were performed with untreated as well as treated clay material. The EXAFS spectra were very similar, and in both cases the data analysis support the presence of a Ni-phyllsilicate phase (see the section on EXAFS and Table 3). This suggests that the purification and conditioning process did not modify the responsible uptake mechanism under the reaction conditions used.

#### *Possible sources of Si in solution required for the neof ormation of phyllsilicates*

In a previous P-EXAFS study, Schlegel *et al.* (2001) observed the neof ormation of a Zn-phyllsilicate phase upon treating hectorite with Zn. Whereas the reaction with Zn required the addition of Si, no Si addition was

needed to form neophyllsilicates in the uptake of Ni onto montmorillonite (this study; Dähn *et al.*, 2002). The Si concentrations in this study (282–527  $\mu\text{M}$ ; Table 1) were comparable to those in the previous P-EXAFS study (177–448  $\mu\text{M}$ , Dähn *et al.*, 2002) and were apparently sufficient for the neof ormation of a Ni-phyllsilicate phase. Dähn *et al.* (2002) speculated that one reason for this distinct difference between the uptake mechanism of Zn on hectorite and Ni on montmorillonite was that the dissolution of  $\text{SiO}_2$  particles in the sorbent material contributed to an increased  $[\text{Si}]_{\text{aq}}$  in the latter case, thus facilitating the neof ormation of Ni-phyllsilicate (this study, Dähn *et al.*, 2002). The TEM study demonstrated the presence of an X-ray amorphous Si-rich phase and thus confirms the above hypothesis. Another Si source observed in this study are halloysite particles. The TEM images suggest that halloysite particles suffer severe surface alteration during the experiments. The observed holes (Figure 7) are interpreted to be caused by partial dissolution of the halloysite particles during the Ni uptake experiment and thus halloysite particles represent a potential source of Al and Si in the system.

## CONCLUSIONS

This study demonstrates that the uptake of Ni onto ‘as-received’ montmorillonite (pH 8,  $[\text{Ni}]_{\text{initial}} = 660 \mu\text{M}$ ) results in the formation of small Ni-rich particles which were identified as neof ormed Ni-phyllsilicates by TEM observations. The finding that neof ormed Ni-phyllsilicates are present in the Ni-doped montmorillonite sample was confirmed by EXAFS measurements. Furthermore, this study confirms previous P-EXAFS studies on the uptake of metal ions on clay minerals (Schlegel *et al.*, 2001; Dähn *et al.*, 2002) and demonstrates that neof ormed phyllsilicates can be observed using TEM. Although TEM observations suggest that purification and conditioning of the clay material is a rather severe treatment, the main Ni uptake process (neof ormation of phyllsilicates) did not change due to the treatment procedures commonly used. This finding implies that the results obtained in metal-uptake experiments with purified and conditioned clay minerals can be extrapolated to predict the fate of heavy metals in clay-rich natural settings.

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